

Review on Mechanical Behavior of High Entropy Alloys

Chair: Prof. Marc A. Meyers

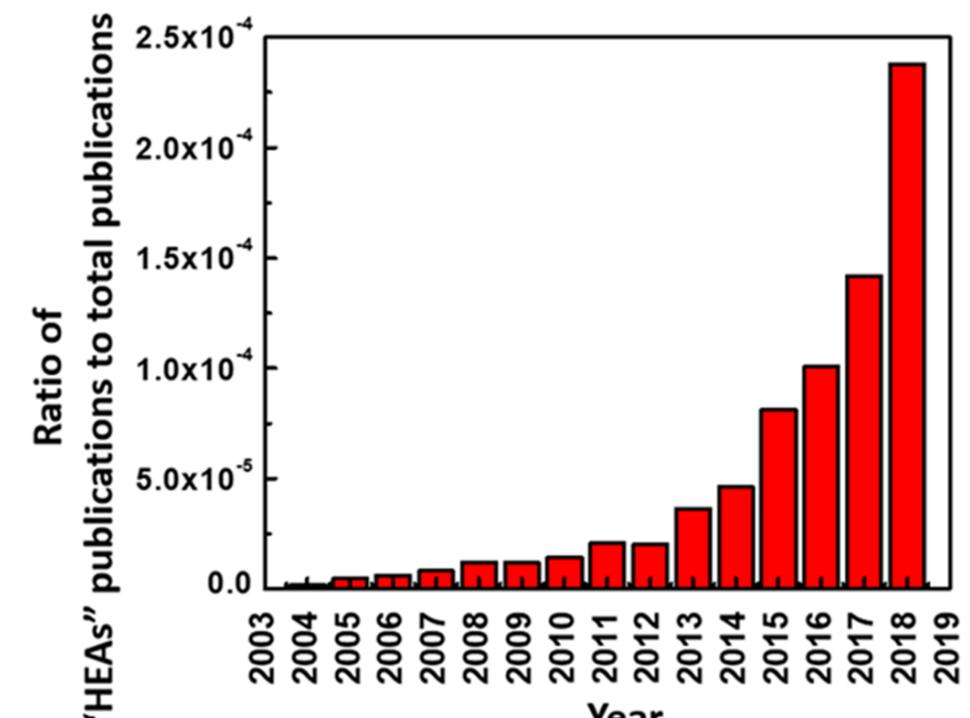
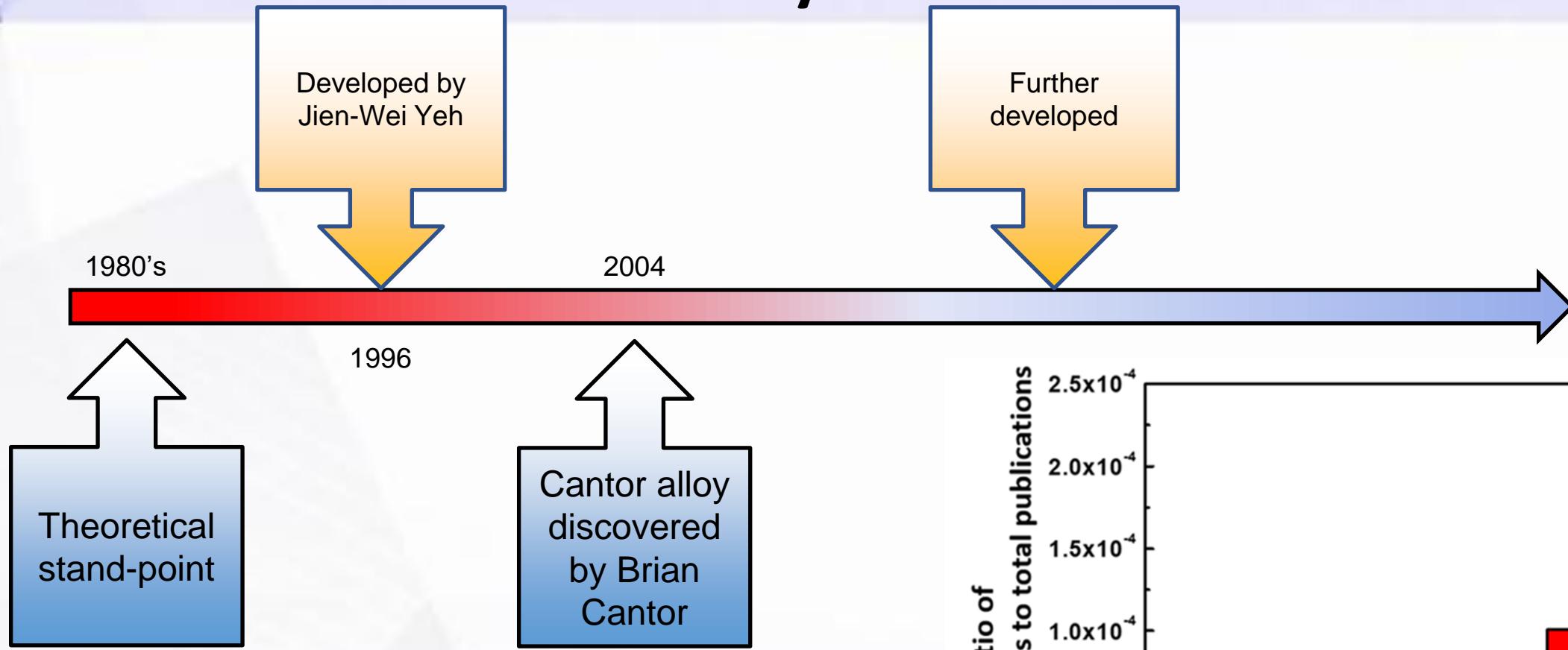
Member: Prof. Shengqiang Cai

Member: Prof. Javier Garay

Member: Prof. Vlado Lubarda

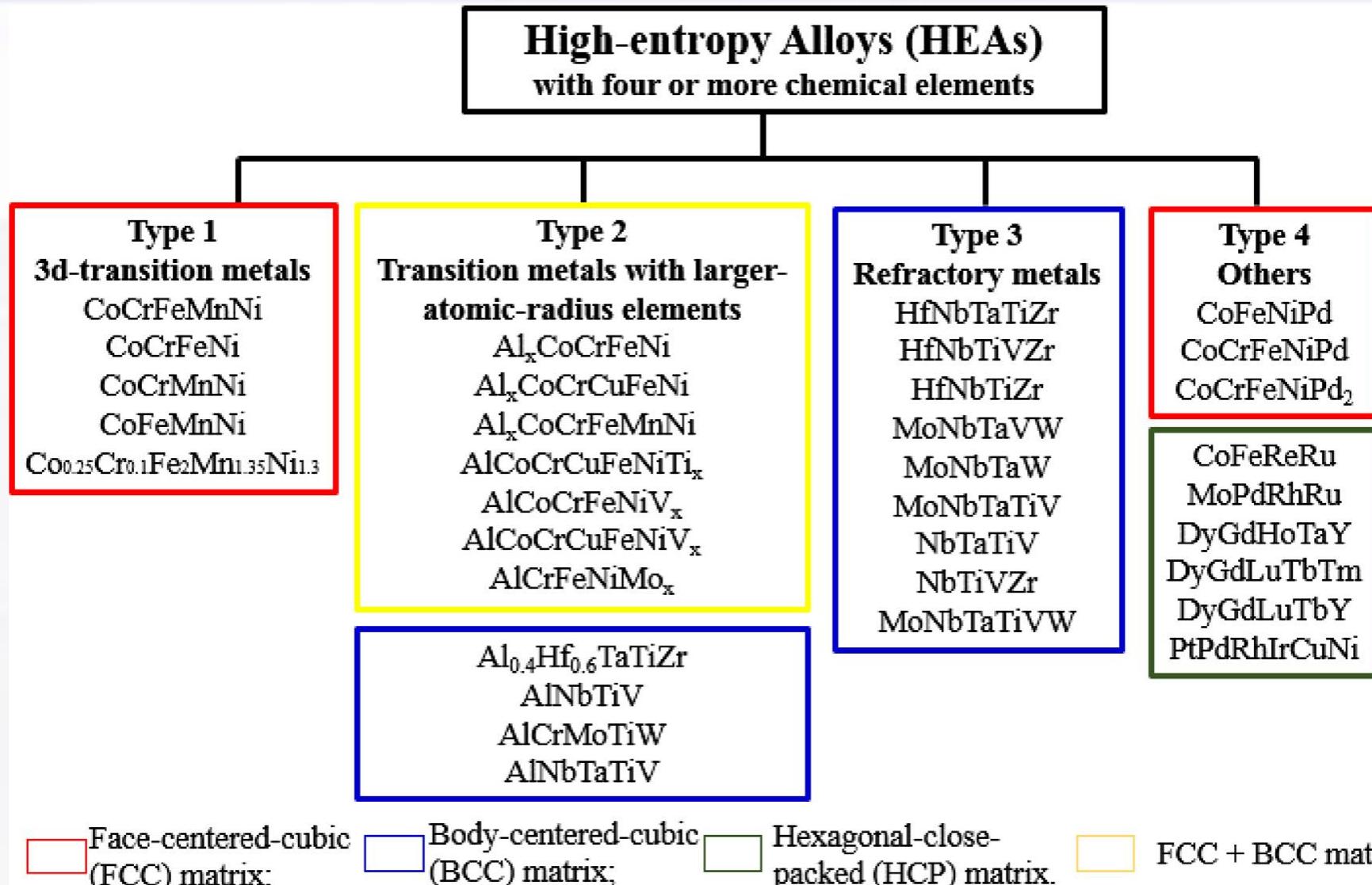
Member: Prof. Vitali Nesterenko

History of HEAs



data from the *Web of Science*²

Classification of HEAs

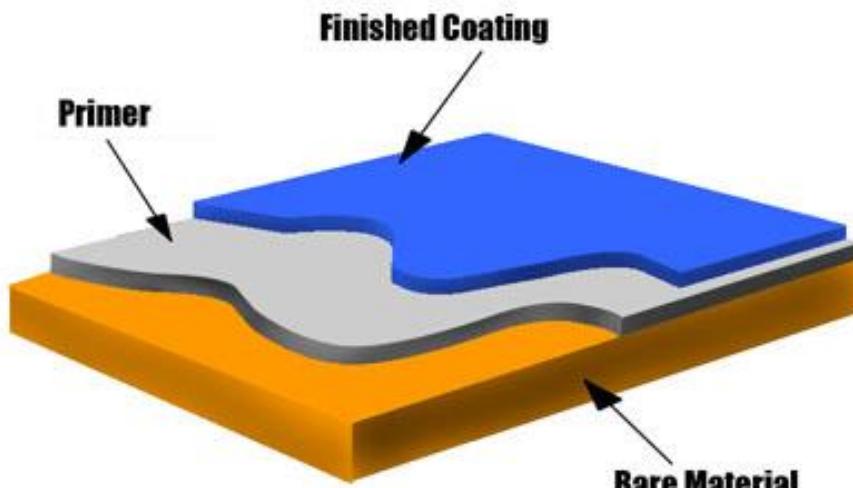
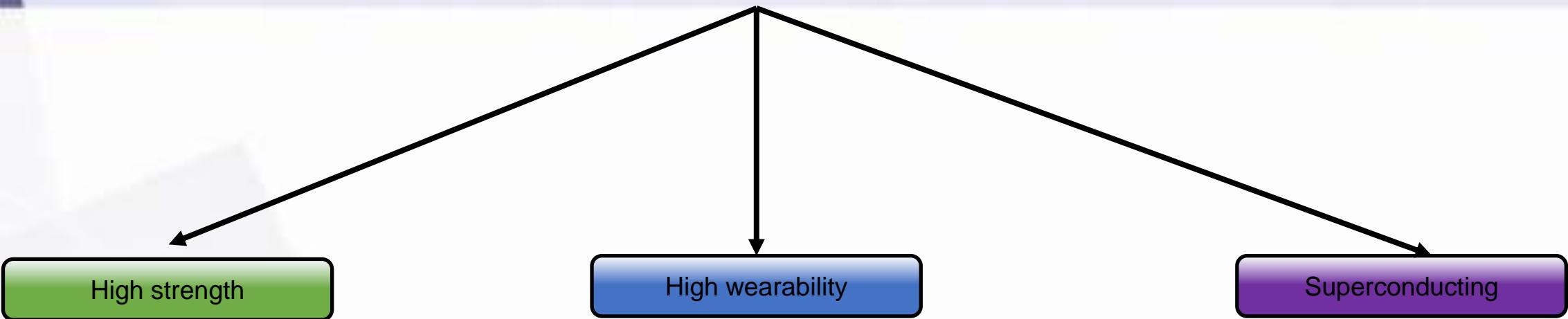


Developed HEA systems

- **Al-Co-Cr-Cu-Fe-Ni alloy system**
- **Derivatives of Al-Co-Cr-Cu-Fe-Ni alloy system:**
 1. **Al-Co-Cr-Cu-Ti alloy system**
 2. **Co-Cr-Fe-Mn-Ni alloy system**
 3. **Al-Cr-Fe-Mn-Ni alloy system**

...
- **Refractory HEA system:**
Refractory elements: Cr, Hf, Mo, Nb, Ta, V, W, Zr

Practical Application



<https://www.caeses.com/industries/case-studies/turbine-blade-optimization-including-scallops-for-a-turbocharger/>

<https://www.globalspec.com/learnmore/materials chemicals adhesives/industrial sealants coatings/industrial coatings>

<https://ricardo.com/news-and-media/news-and-press/ricardo-performance-products-achieves-key-aerospace-manufacturing-certification>

Core effects of HEAs

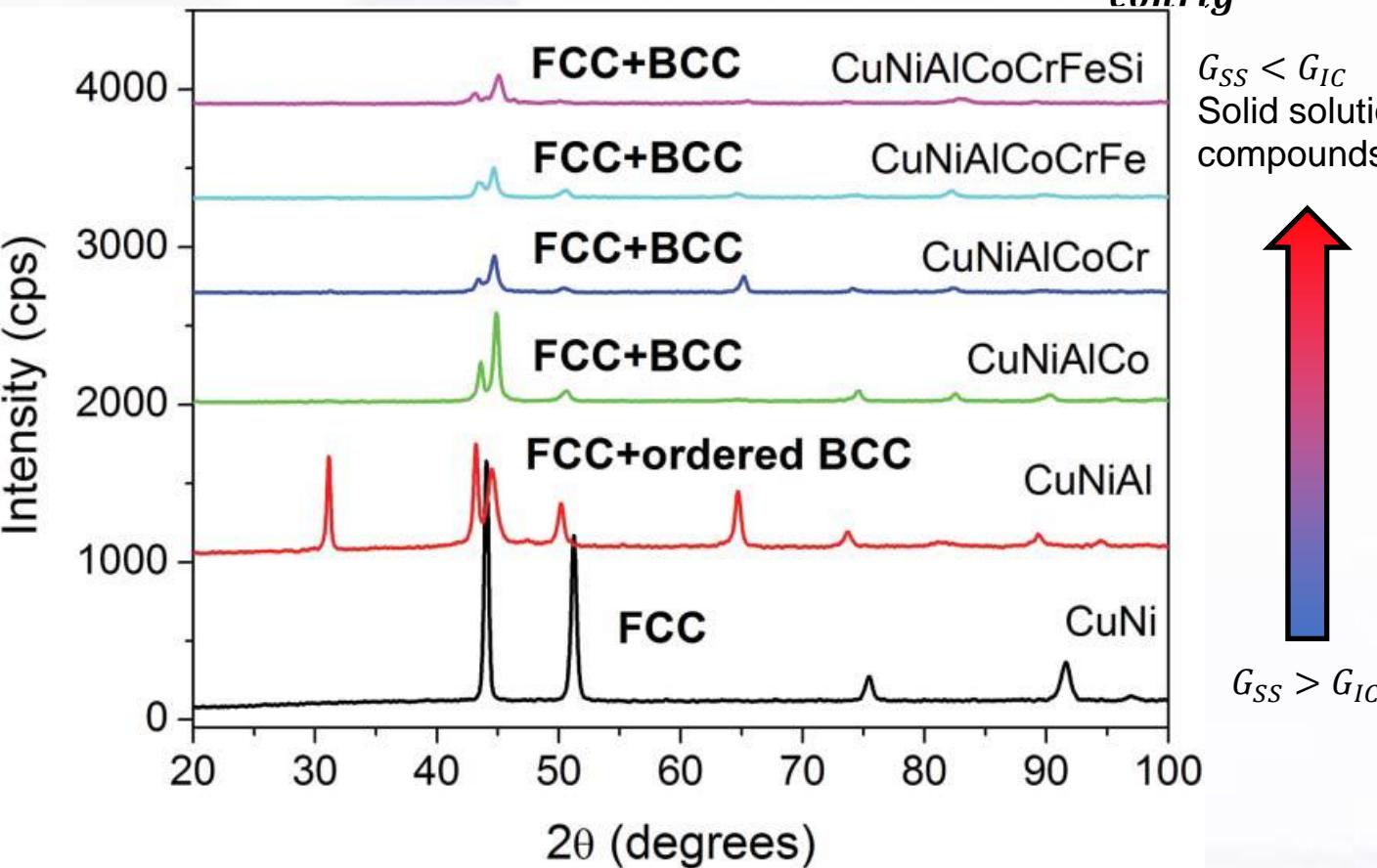
- 1. High entropy effect
- 2. Sluggish diffusion effect (?)
- 3. Severe lattice-distortion effect
- 4. Cocktail effect
- 5. Short-range ordering

High entropy effect

$$G = H - TS, \Delta S_{config} = R \ln N \text{ (N is number of elements in equimolar ratio)}$$

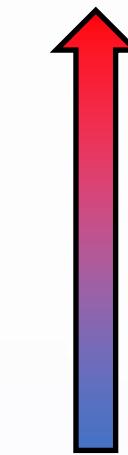
$$\Delta G_{mix} = \Delta H_{mix} - T\Delta S_{mix} \left(-22 \leq \Delta H_{mix} \leq \frac{7 \text{ kJ}}{\text{mol}} \right) (\Delta S_{mix} > 1.6R)$$

$S_{config} > 1.6R$

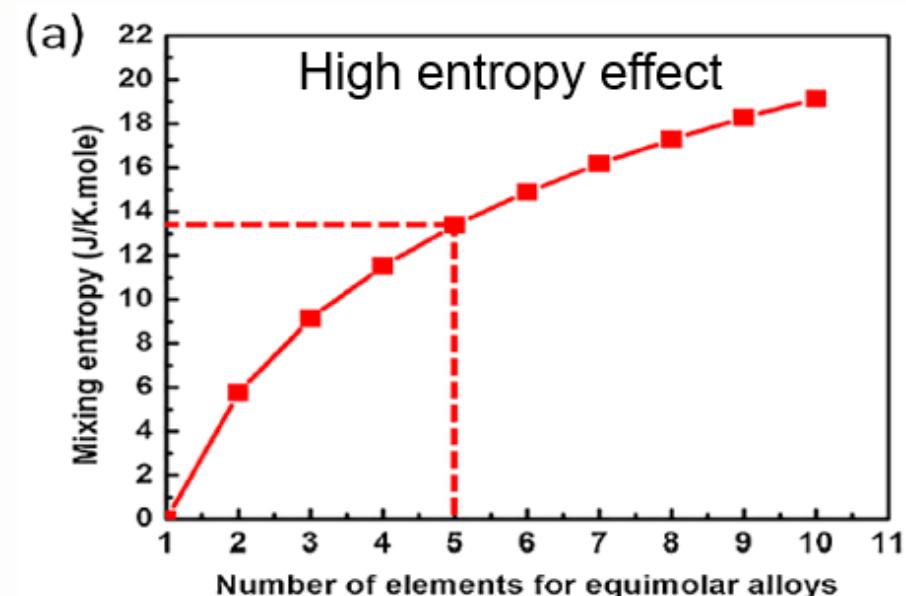


XRD pattern of different multi-component alloy

$G_{SS} < G_{IC}$
Solid solution over intermetallic compounds

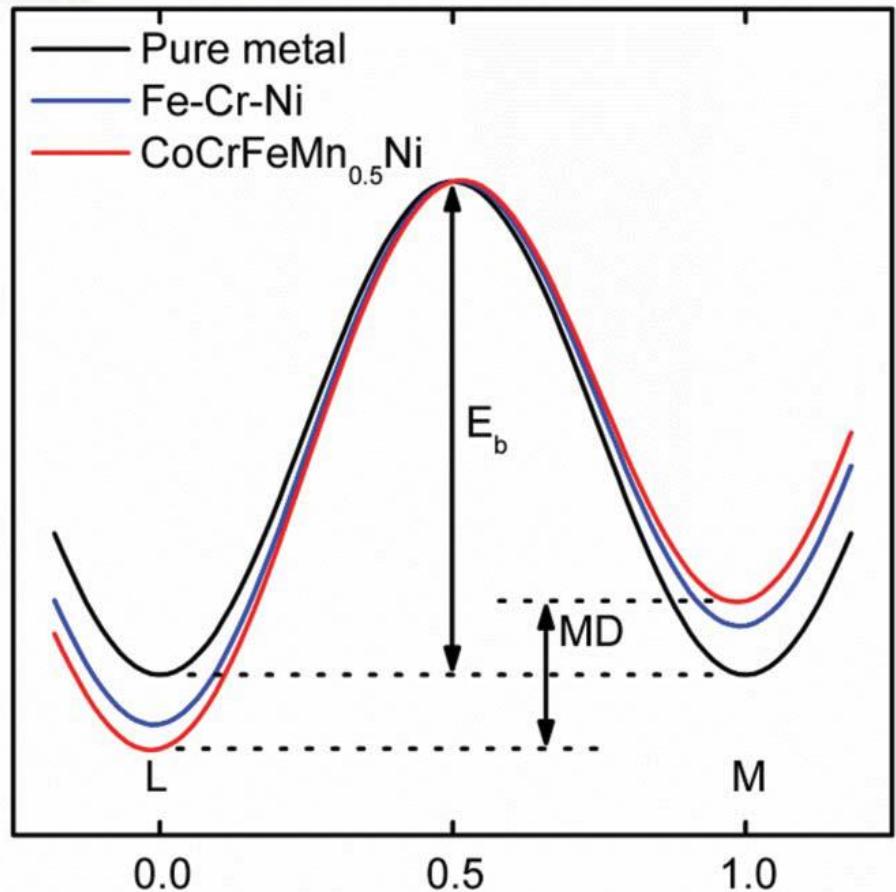


$G_{SS} > G_{IC}$



Sluggish diffusion effect

Potential Energy



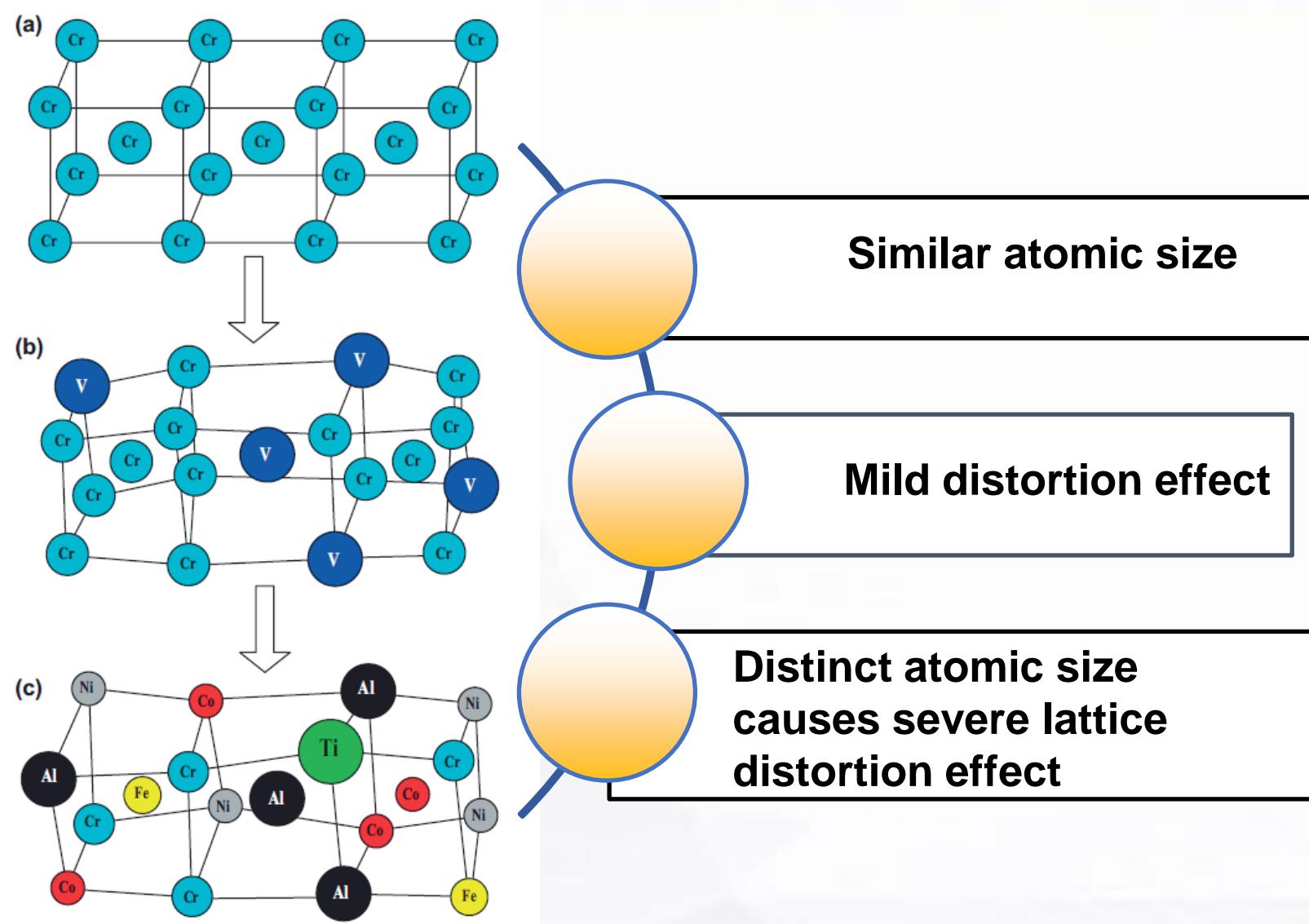
lattice potential energy (LPE) and mean difference
when a Ni atom migrates in pure metal

**Higher energy to overcome barrier for
atoms to migrate from site L to site M**

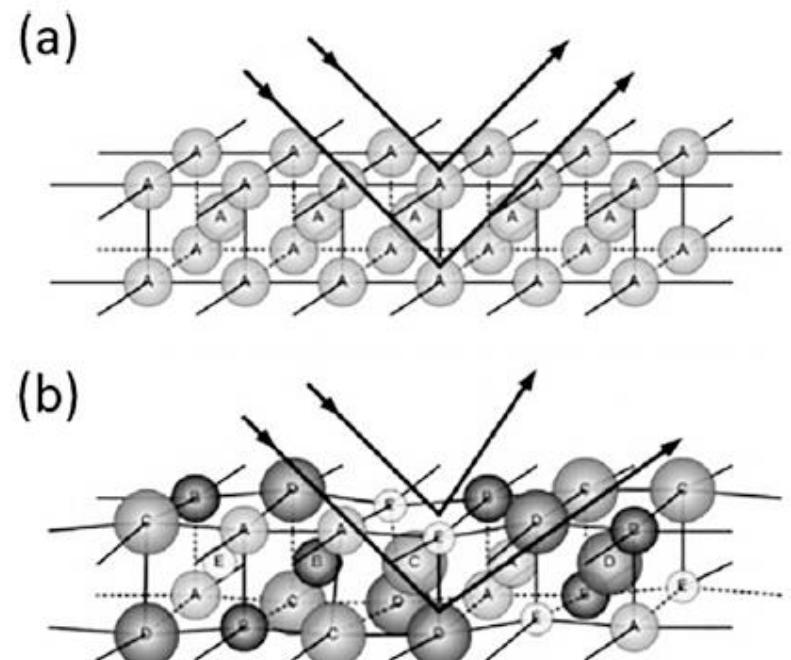
Solute	System	D_0 ($10^{-4} \text{ m}^2/\text{s}$)	Q (kJ/mol)	$T_m(T_s)$ (K)	Q/T_m	D_{T_m} ($10^{-13} \text{ m}^2/\text{s}$)
Ni	CoCrFeMnNi	19.7	317.5	1607	0.1975	0.95
	FCC Fe	3	314	1812	0.1733	2.66
	Co	0.43	282.2	1768	0.1596	1.98
	Ni	1.77	285.3	1728	0.1651	4.21
	Fe-15Cr-20Ni	1.5	300	1731	0.1733	1.33
	Fe-15Cr-45Ni	1.8	293	1697	0.1727	1.73
	Fe-22Cr-45Ni	1.1	291	1688	0.1724	1.09
	Fe-15Cr-20Ni-Si	4.8	310	1705	0.1818	1.53

Diffusion parameters for Ni

Lattice distortion effect

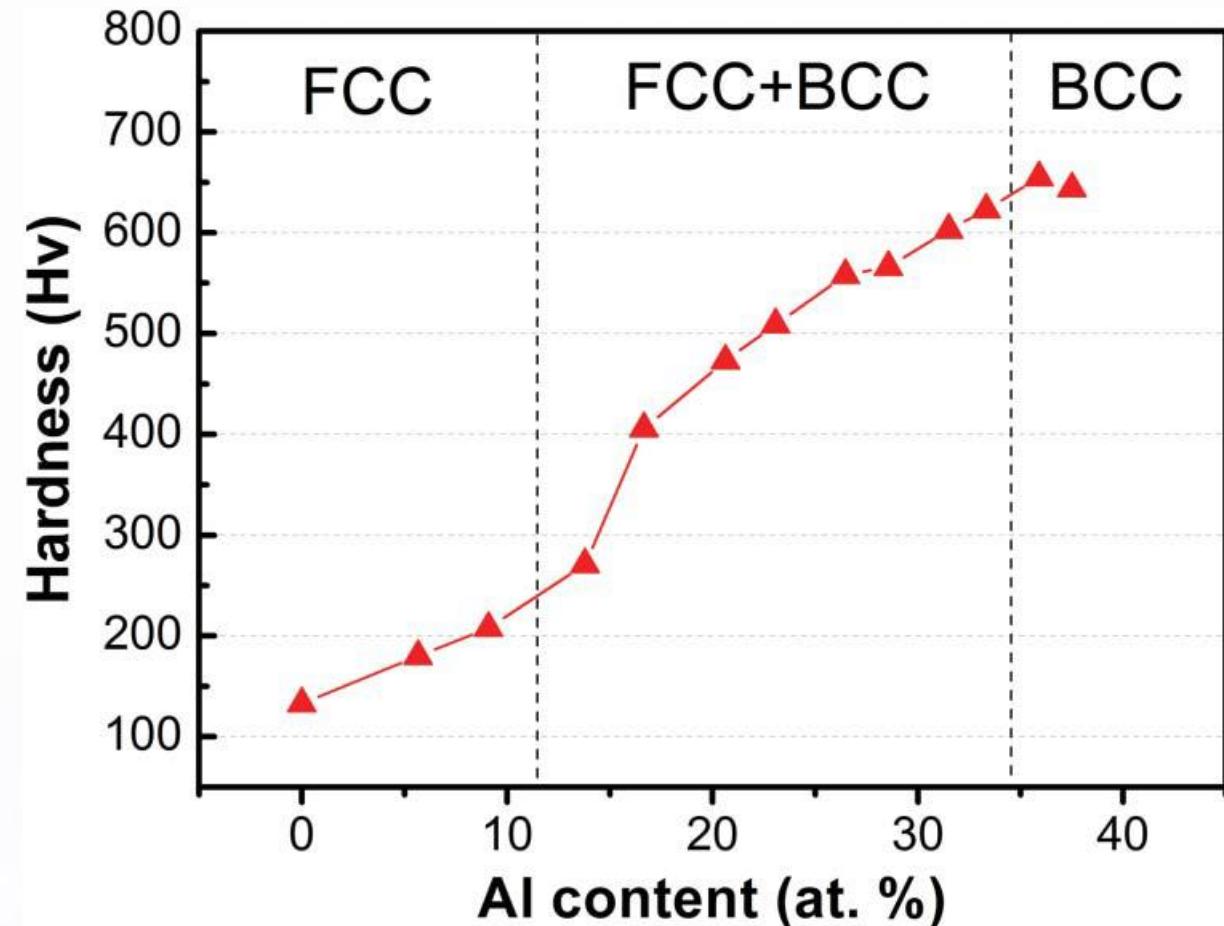
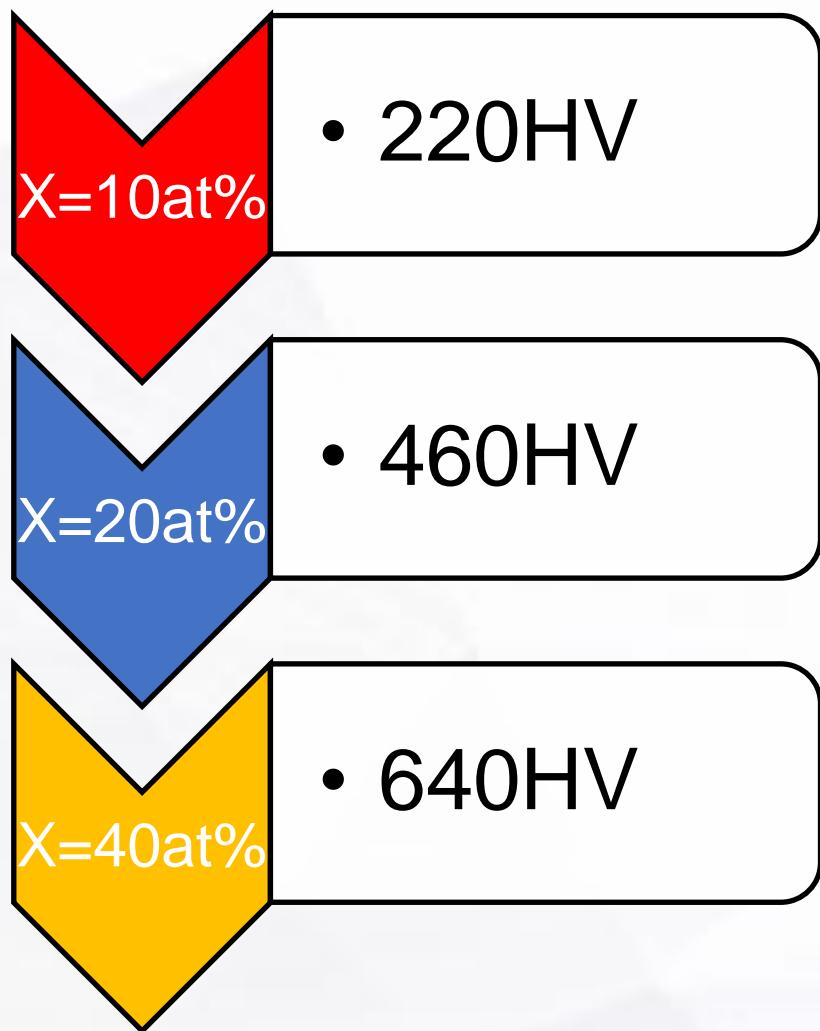


Weakened XRD intensity



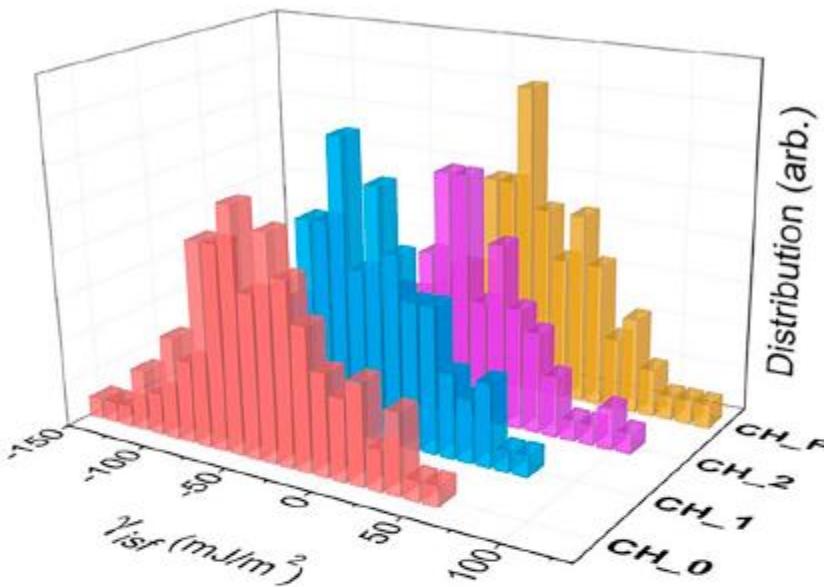
Cocktail effect

$\text{Al}_x\text{CoCrCuFeNi}$ HEA
Change atomic ratio of Al

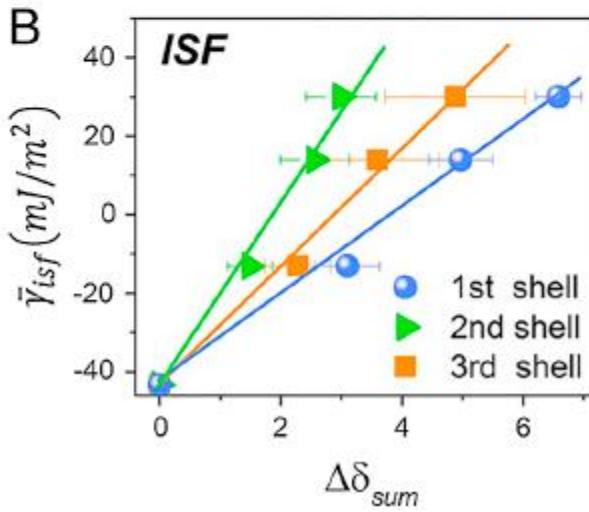


Short-range ordering

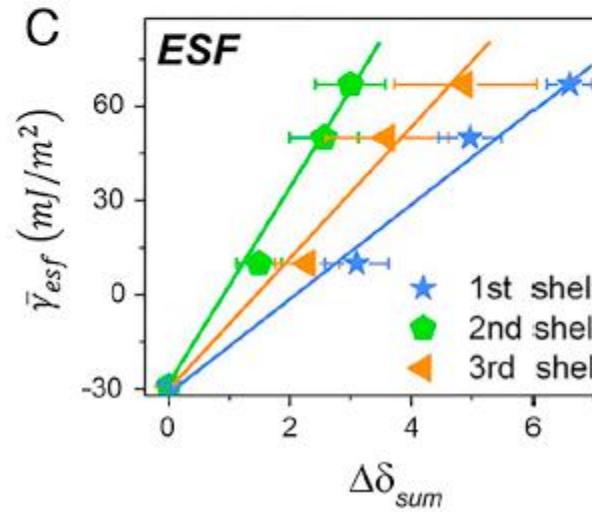
A



B



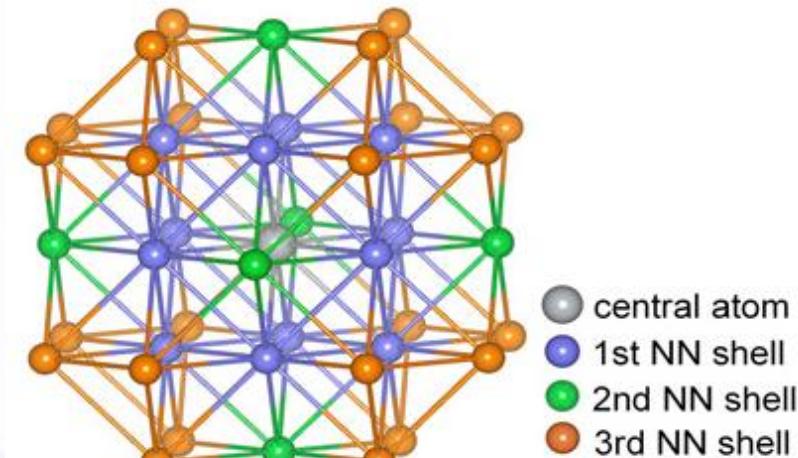
C



$\bar{\gamma}_{isf}$ (Intrinsic stacking fault energy):
-42.9 to 30 mJ/m^2

$\bar{\gamma}_{esf}$ (Extrinsic stacking fault energy):
-27.8 to 66 mJ/m^2

VEC, lattice distortion, d-electron density

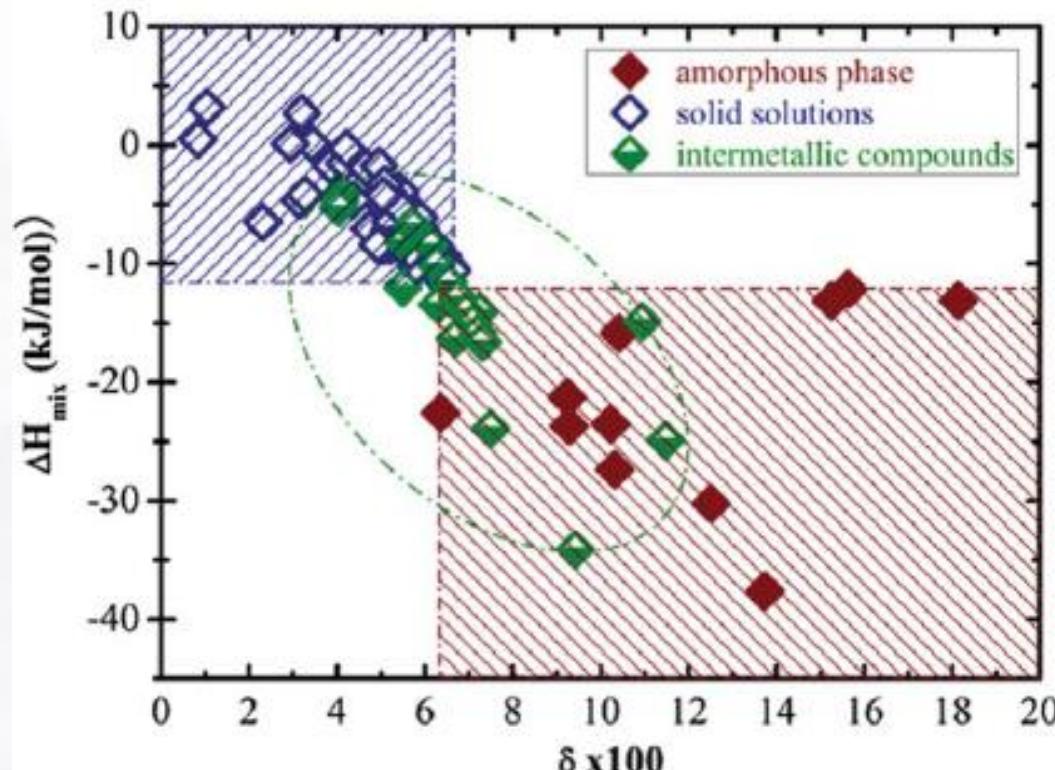


Stacking fault energy and schematic illustration of first three nearest-neighbour shells in FCC

11

Hume-Rothery Rule

1. Mismatch= $\left(\frac{r_{\text{solute}} - r_{\text{solvent}}}{r_{\text{solvent}}} \right) \times 100 \leq 15\%$
2. Crystal structure rule
3. Valence rule
4. Electronegativity rule



$$\delta = \sqrt{\sum_{i=1}^N c_i \left(1 - \frac{r_i}{\bar{r}}\right)^2}$$
$$\bar{r} = \sum_{i=1}^n c_i r_i$$

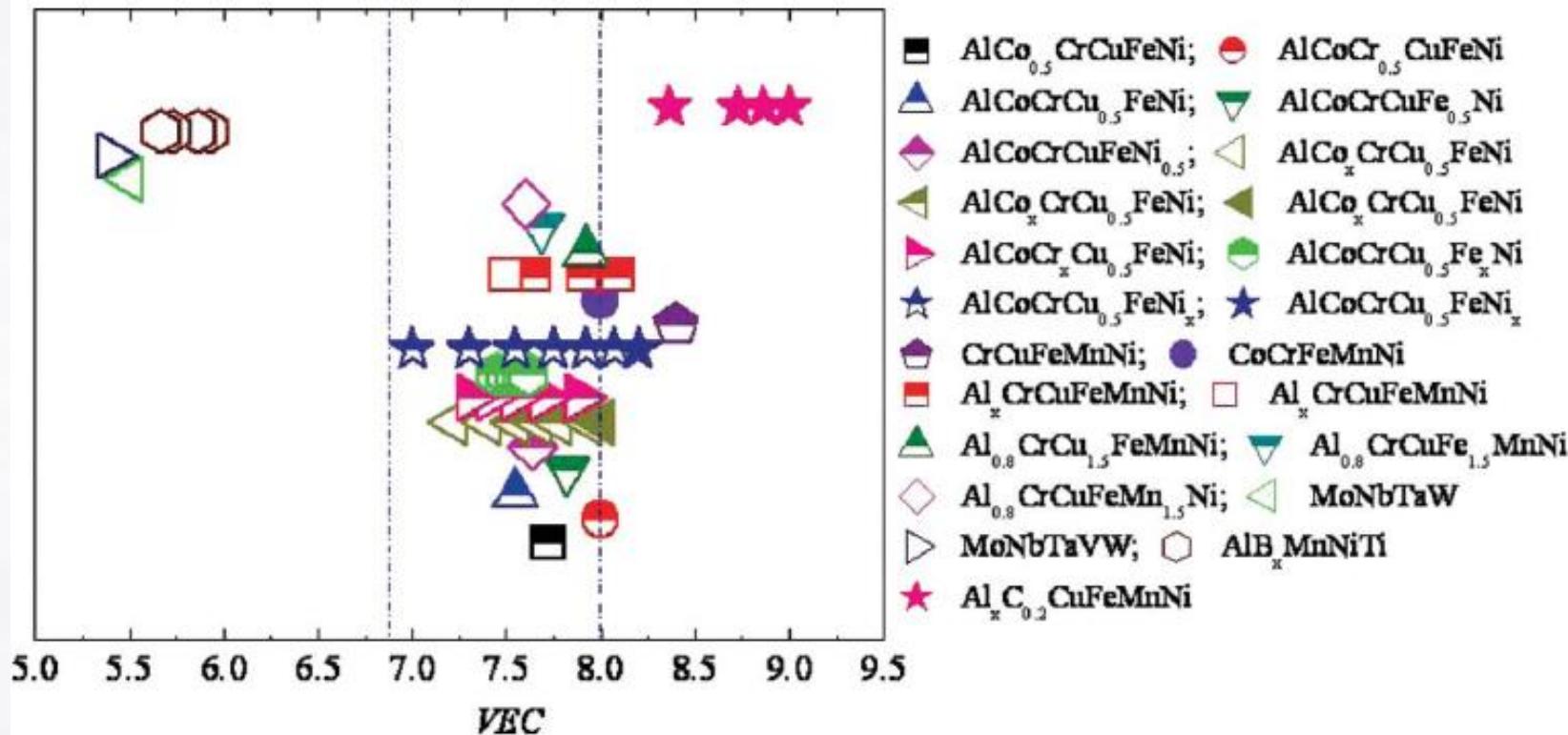
FCC or BCC HEA

(VEC) Valence electron concentration

VEC>8: single FCC

VEC<6.8: single BCC

6.8<VEC<8: co-existence of BCC & FCC



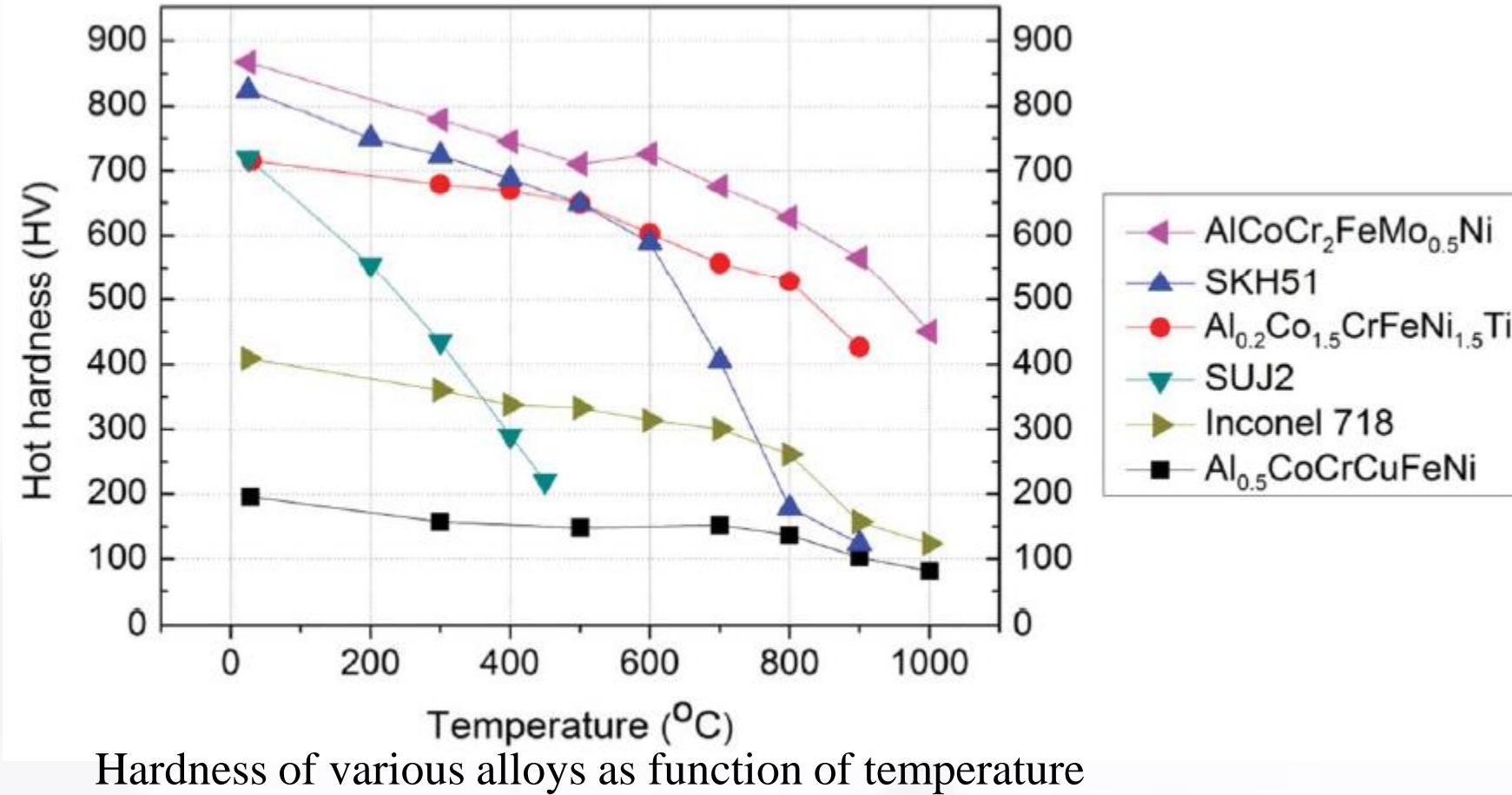
Relationship between VEC and BCC, FCC phase stability

High temperature performance

Resistance to thermal softening: T_T

Softening coefficient of conventional alloy and $\text{AlCoCr}_x\text{FeMo}_{0.5}\text{Ni}$

Alloy	Softening coefficient above T_T
Cr-1.5	-2.32E-4
Cr-2	-1.66E-4
T-800	-3.12E-4
In 718	-3.55E-4

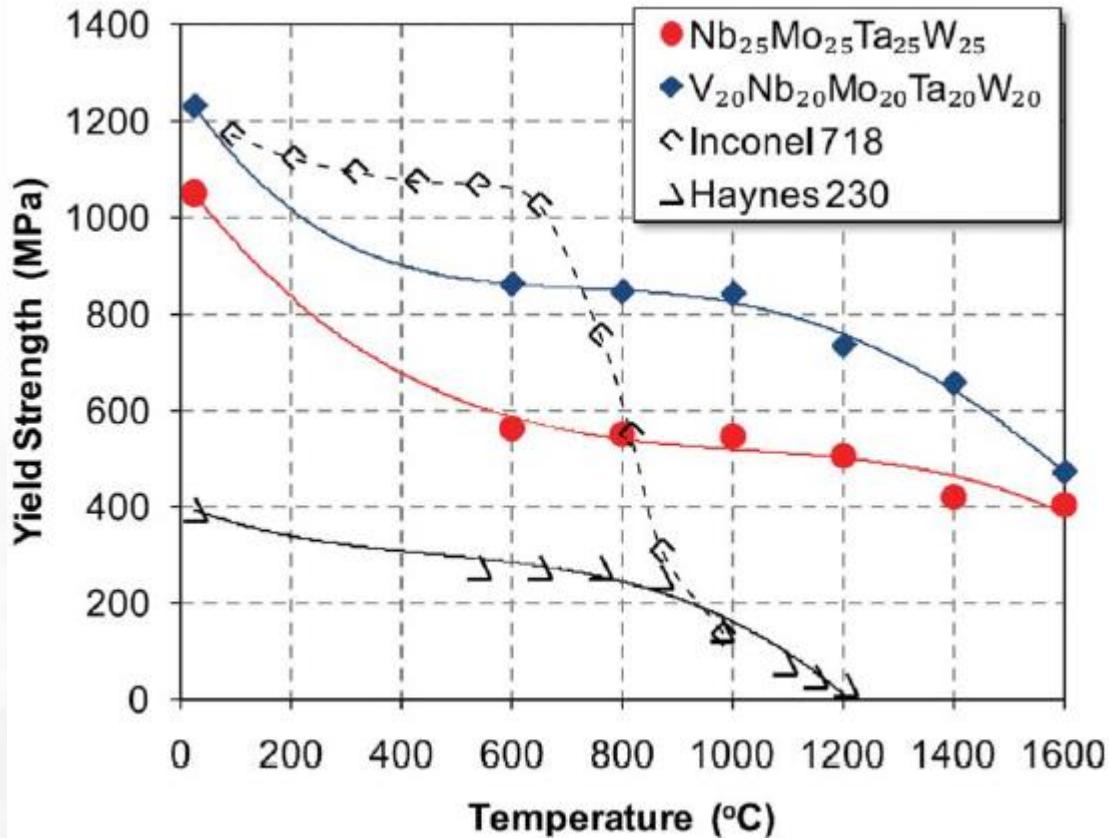


1. Chuang MH, Tsai MH, Wang WR, Lin SJ, Yeh JW. Microstructure and wear behavior of $\text{Al}_x\text{Co}1.5\text{CrFeNi}1.5\text{Ti}_y$ high-entropy alloys. *Acta Mater.* 2011;59:6308–6317.

2. Hsu CY, Juan CC, Wang WR, Sheu TS, Yeh JW, Chen SK. On the superior hot hardness and softening resistance of $\text{AlCoCr}_x\text{FeMo}_{0.5}\text{Ni}$ high-entropy alloys. *Mater Sci Eng A.* 2011;528:3581–3588.

High temperature performance

Low diffusion rate & High transition temperature & reduced driving force to eliminate defects



Yield strength as functions of temperature for the $\text{Nb}_{25}\text{Mo}_{25}\text{Ta}_{25}\text{W}_{25}$ and $\text{V}_{20}\text{Nb}_{20}\text{Mo}_{20}\text{Ta}_{20}\text{W}_{20}$ alloys and two superalloys.

Synthesis Methods of HEA

- Liquid state: Arc melting, induction melting, Bridgman solidification
- Solid state: High-energy ball mill
- Gas state: Sputtering or molecular beam epitaxy (MBE)
- Other methods: thermal spray, laser cladding, electrodeposition

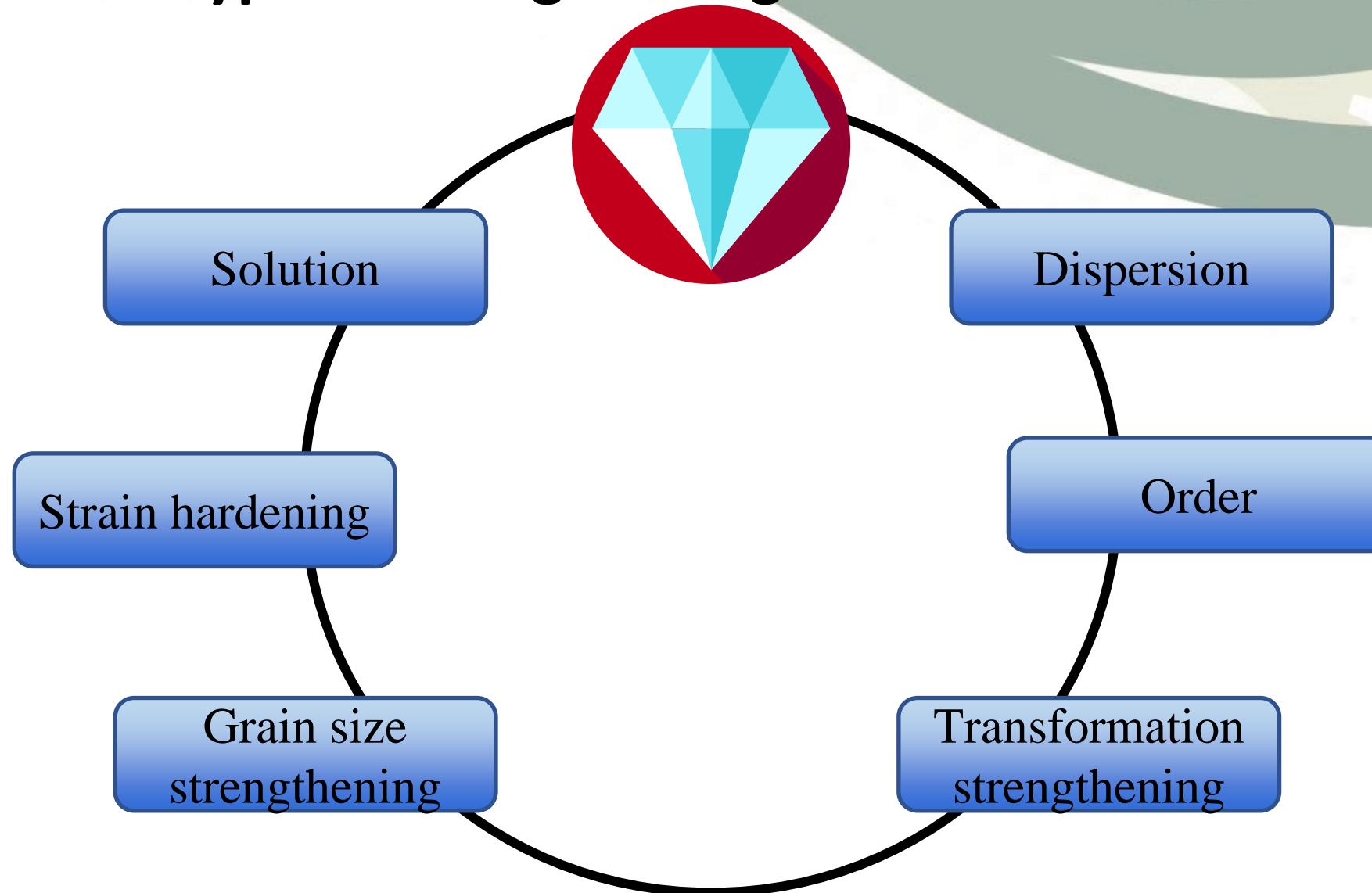
Mechanical properties of HEA

- Hardness and strength
- Wear and fatigue
- Fracture
- Creep
- Compression and tensile deformation

Strengthening mechanism

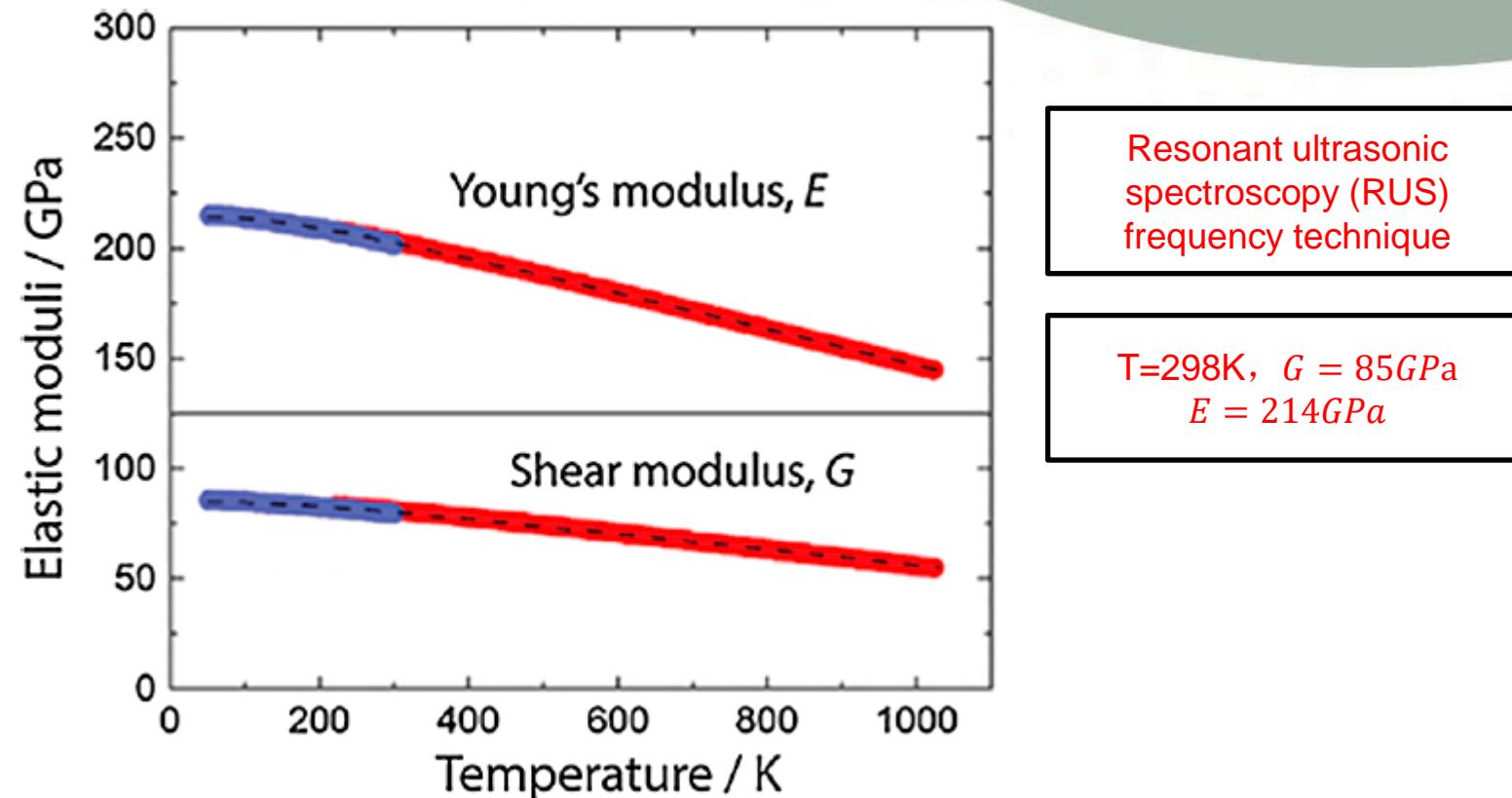
1. Solid solution strengthening
2. Grain size strengthening
3. Twinning
4. Etc.

Typical strengthening mechanism of HEAs



Elastic constant of HEA

- Temperature dependence of elastic modulus G & E of Cantor alloy
 - Normal behavior



Resonant ultrasonic spectroscopy (RUS)
frequency technique

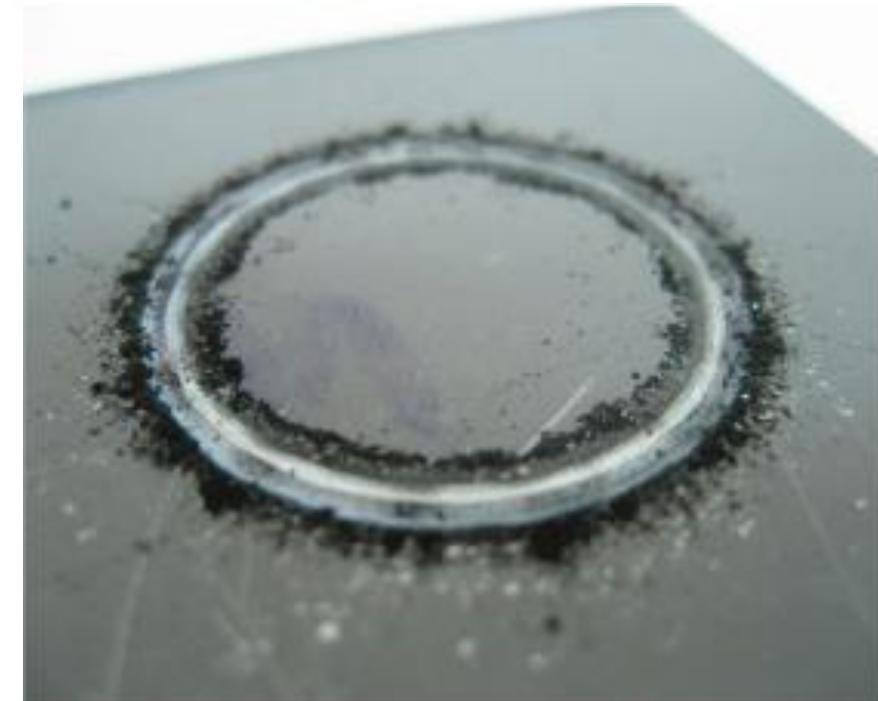
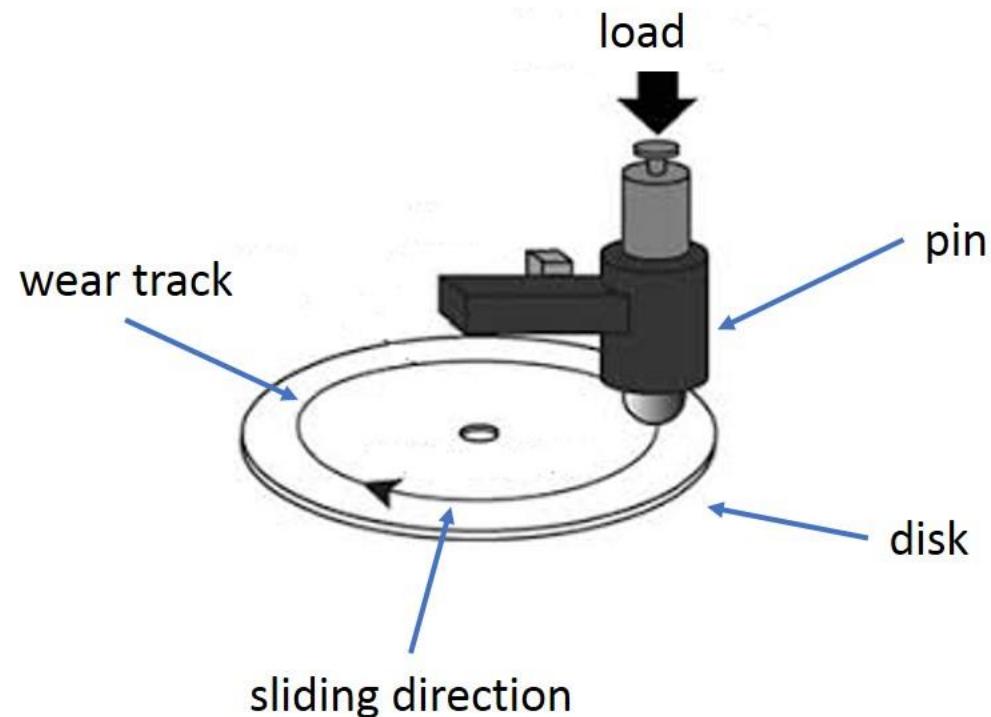
T=298K, $G = 85\text{GPa}$
 $E = 214\text{GPa}$

Calculation of E & G of Cantor alloy (Varshni expression): $G = 85 - 16/(e^{\frac{448}{T}} - 1)$, $E = 214 - 35/(e^{\frac{416}{T}} - 1)$

Wear performance of HEAs

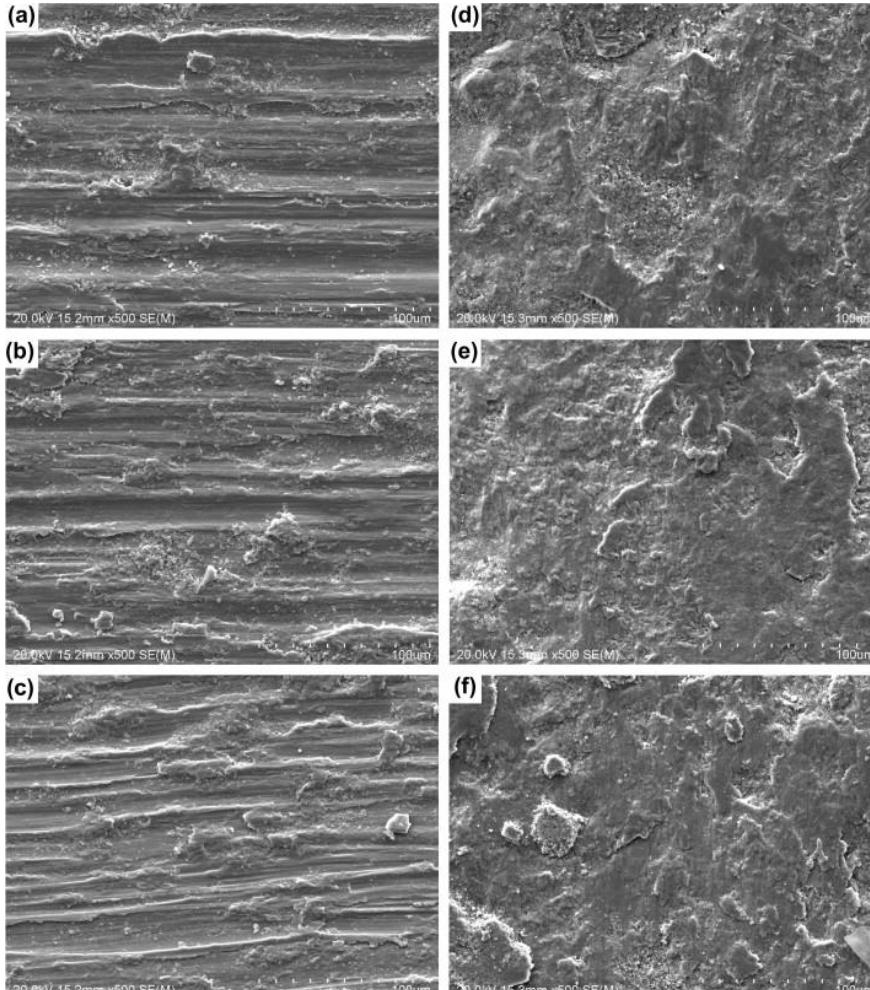
$$W_r = \frac{L}{\Delta V}$$

ΔV = Volume loss, L = sliding distance



Pin on disk test

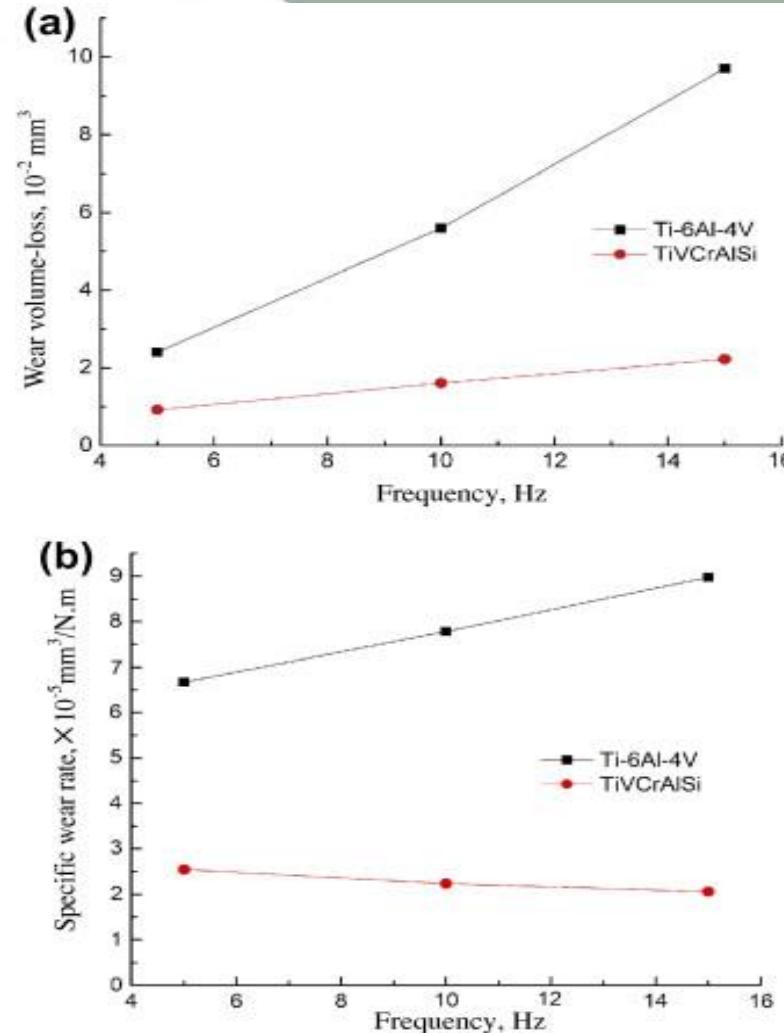
Wear performance of HEAs



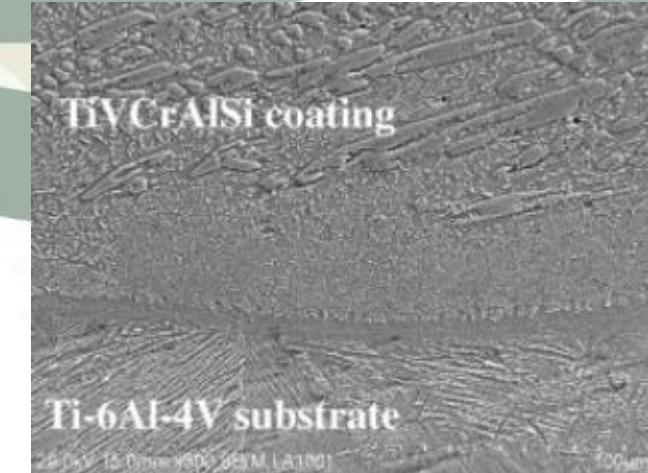
Adhesive wear & severe abrasive wear

Surface morphology of two alloys

Oxidation, less weight loss (Heat generate)



Volumetric loss vs. Frequency



Hardness & wear resistance:

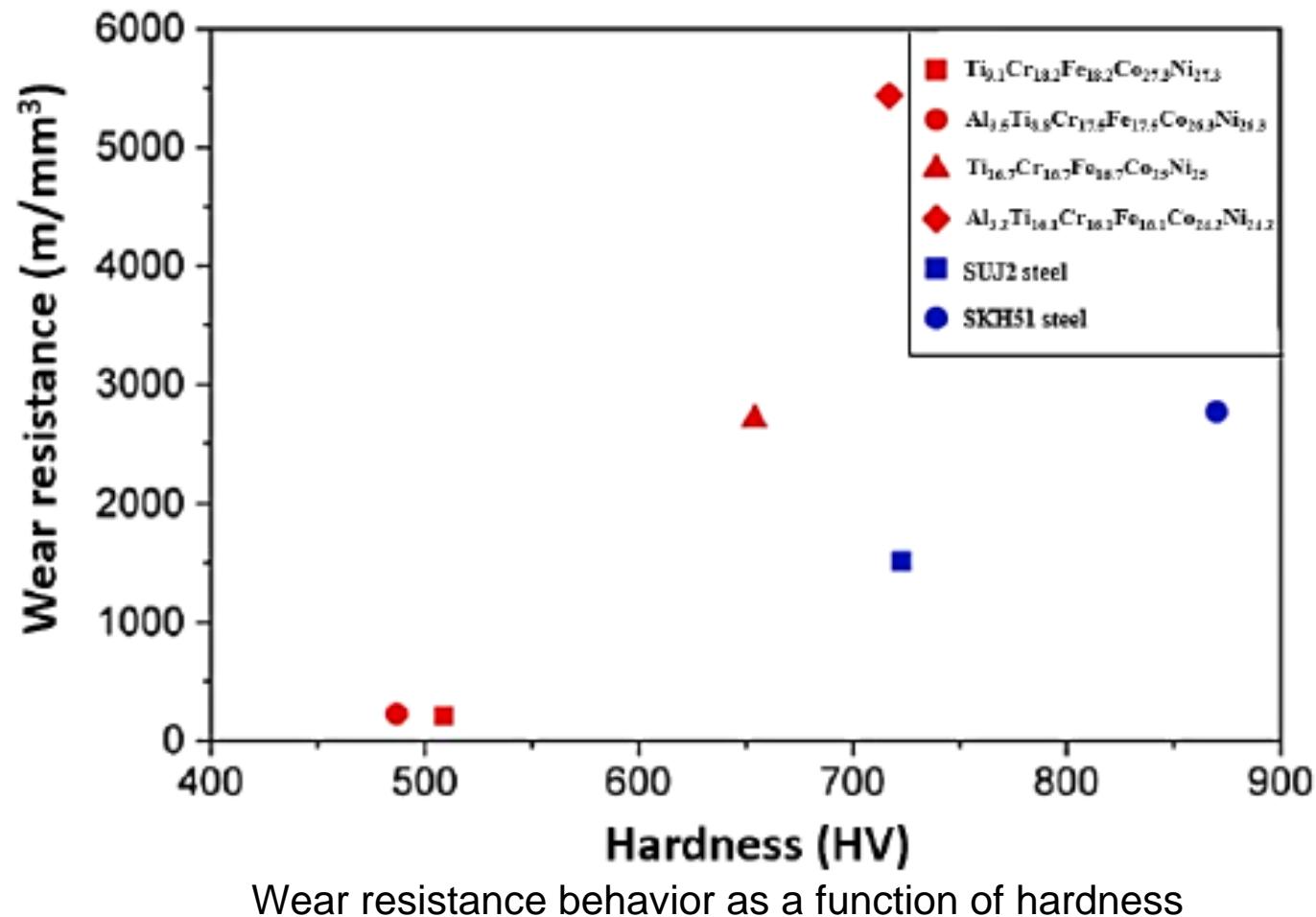
$(\text{Ti},\text{V})_5\text{Si}_3$ and bcc phase solid solution

Thermal conductivity & wear resistance

Friction coefficient

Wear performance of HEAs

HEA is a promising candidate as anti-wear alloys
due to excellent work hardening ability



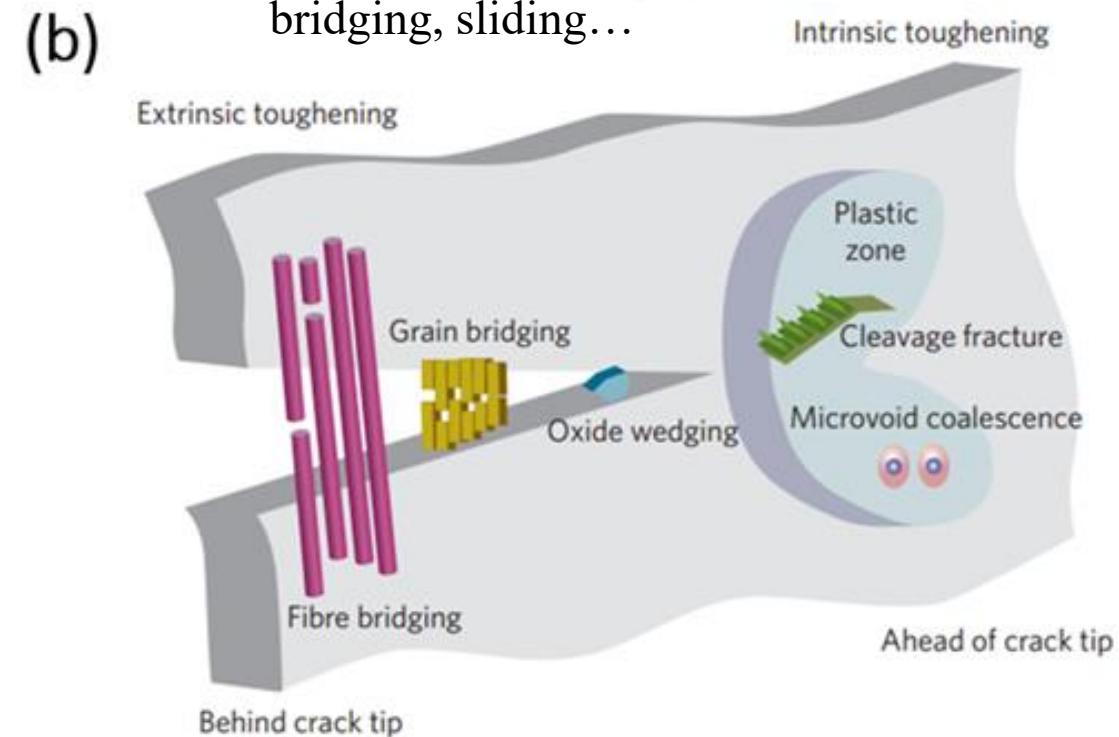
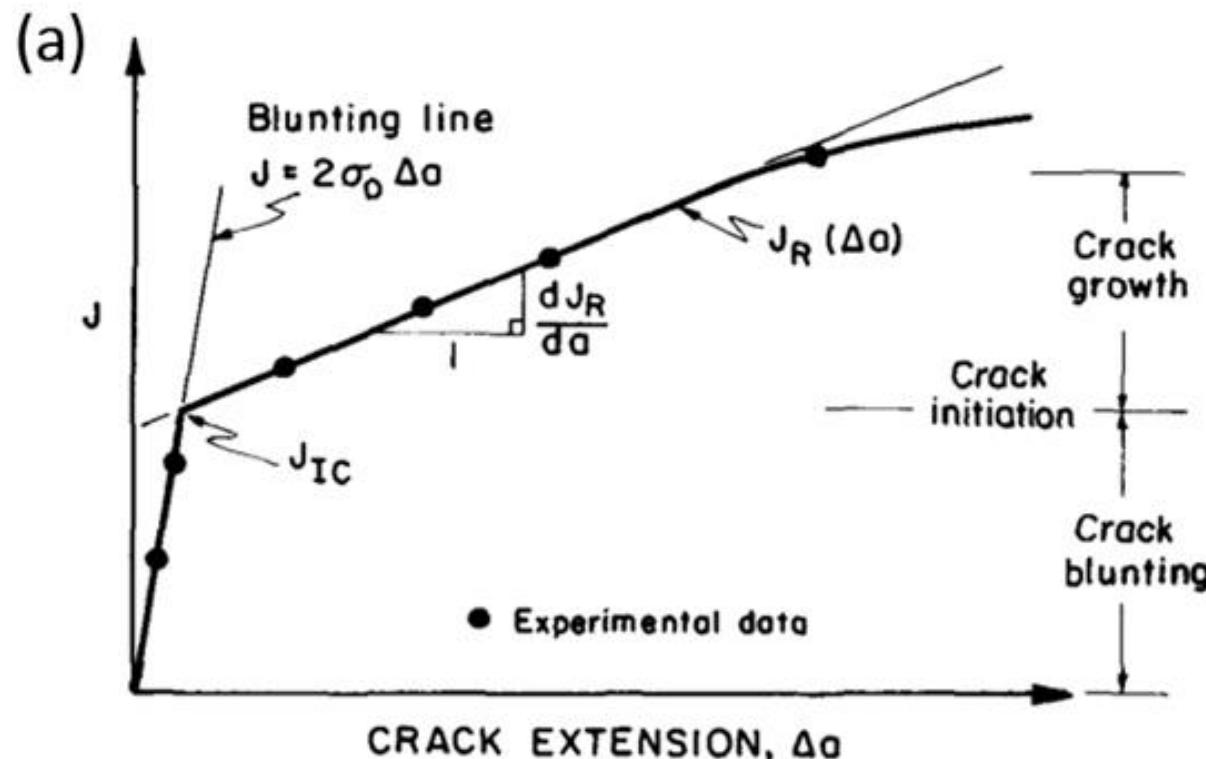
Fracture toughness

Stress intensity factor: K_{1c}

$$J_{1c} = \frac{K_{1c}^2}{E'} = \lambda \delta_{1c} \sigma_0$$

$$E' = E \text{ (Plane stress)}$$

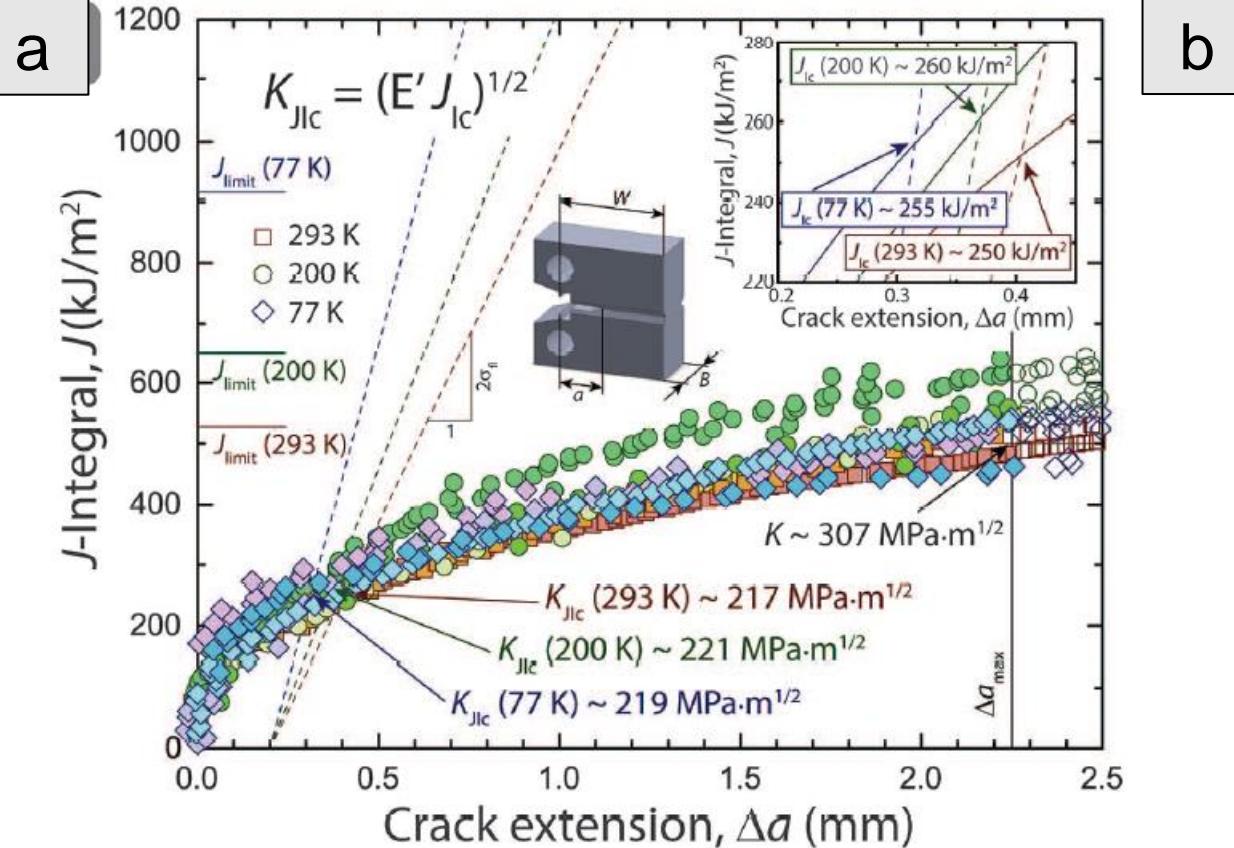
$$E' = \frac{E}{1 - \nu} \text{ (Plane strain)}$$



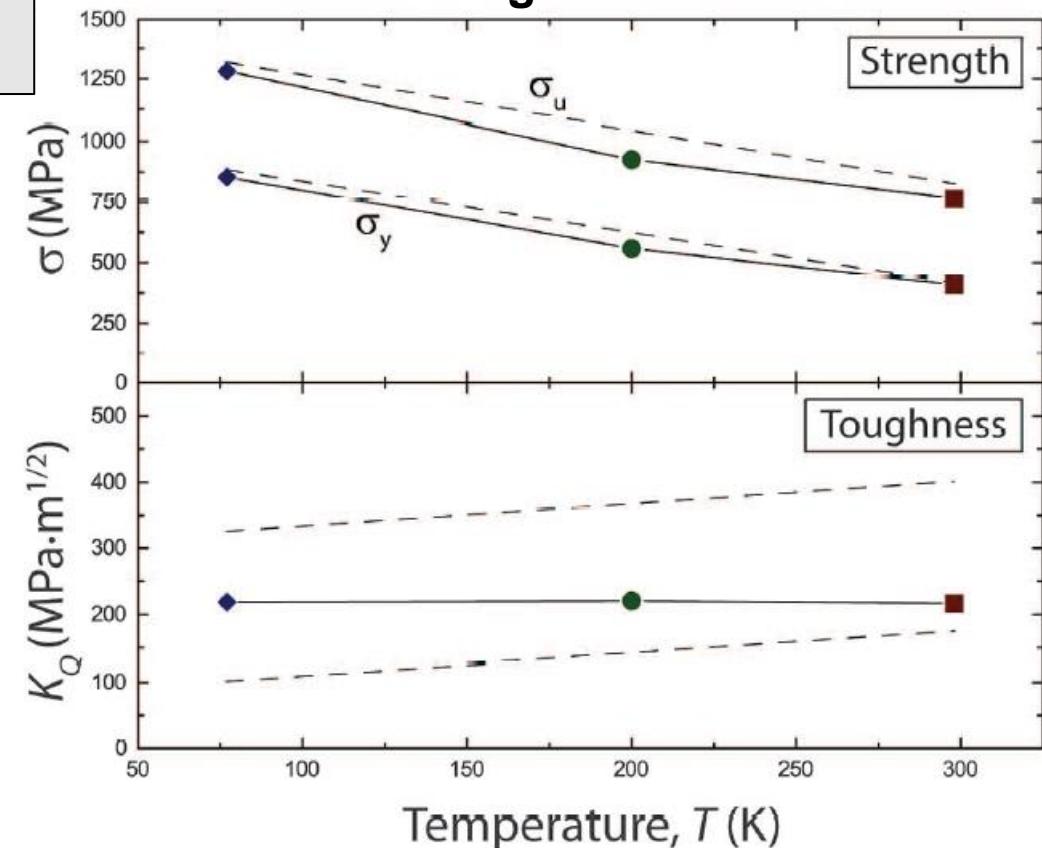
Fracture toughness

Cantor alloy

77K	$J_{1c} = 255 \text{ kJ/m}^2$	$K_{1c} = 219 \text{ MPa} \cdot \text{m}^{\frac{1}{2}}$
200K	$J_{1c} = 260 \text{ kJ/m}^2$	$K_{1c} = 221 \text{ MPa} \cdot \text{m}^{\frac{1}{2}}$
293K	$J_{1c} = 250 \text{ kJ/m}^2$	$K_{1c} = 217 \text{ MPa} \cdot \text{m}^{\frac{1}{2}}$



Temperature dependence of strength and toughness

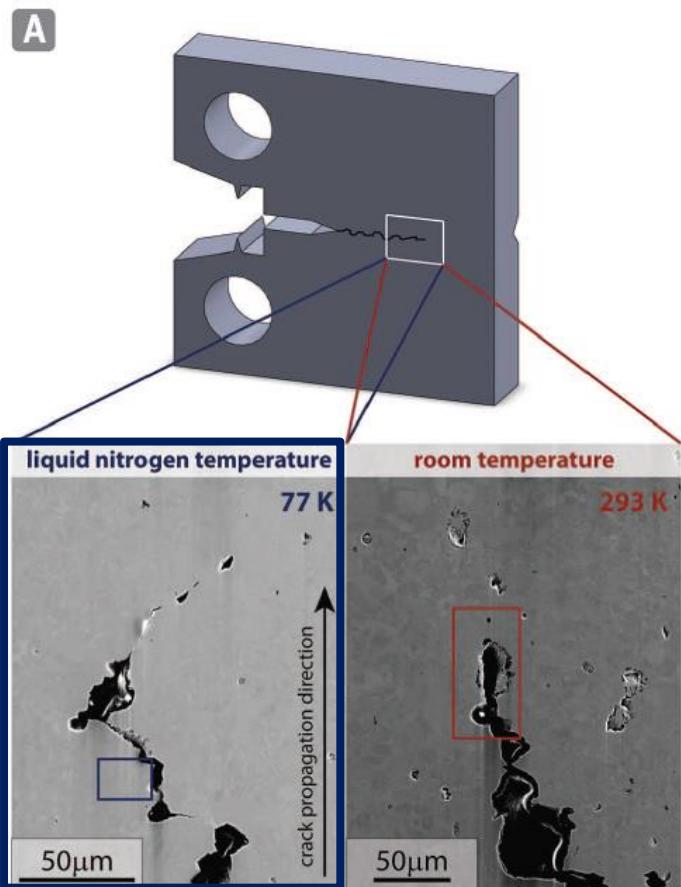
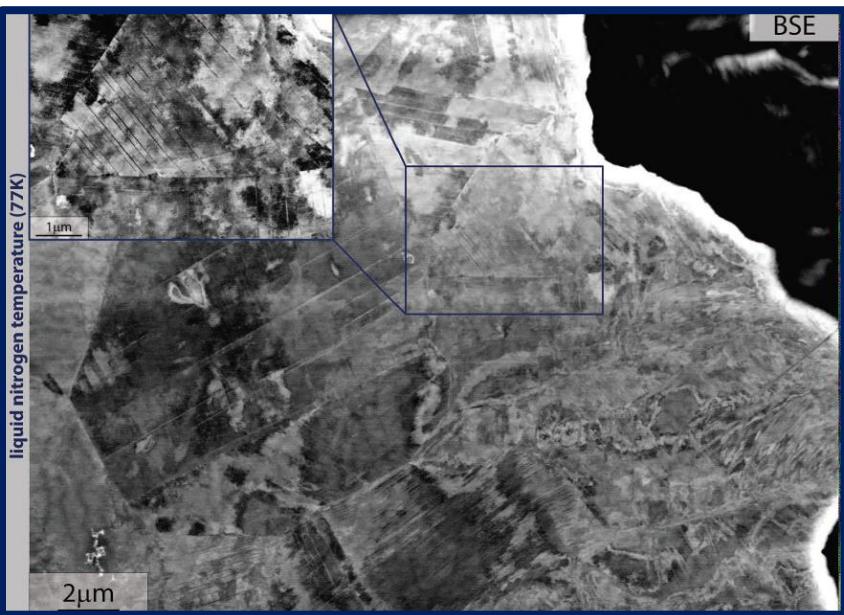


Fracture toughness

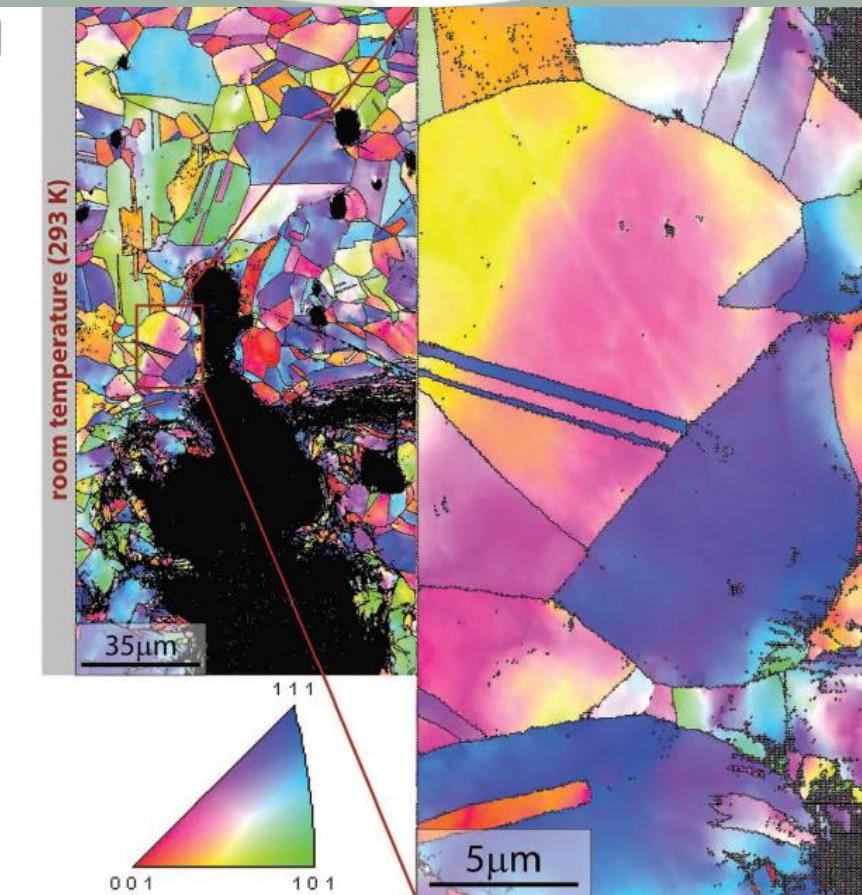
Annealing
twins

Dislocation

Nanotwins



SEM images of crack tip



EBSD images of crack tip

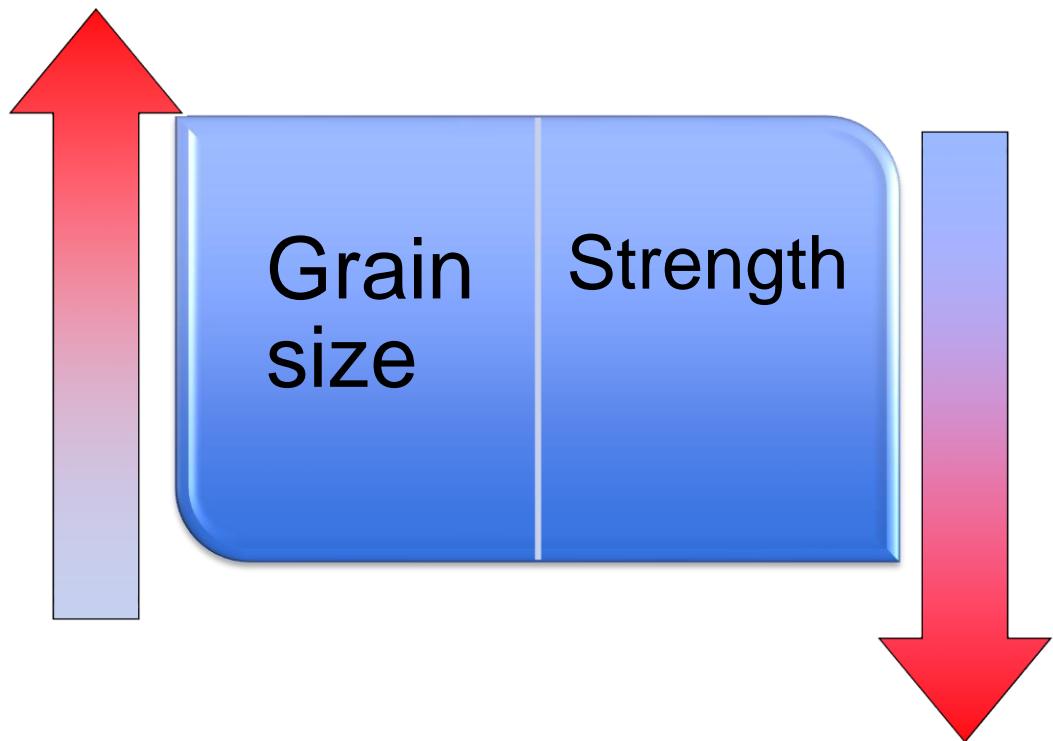
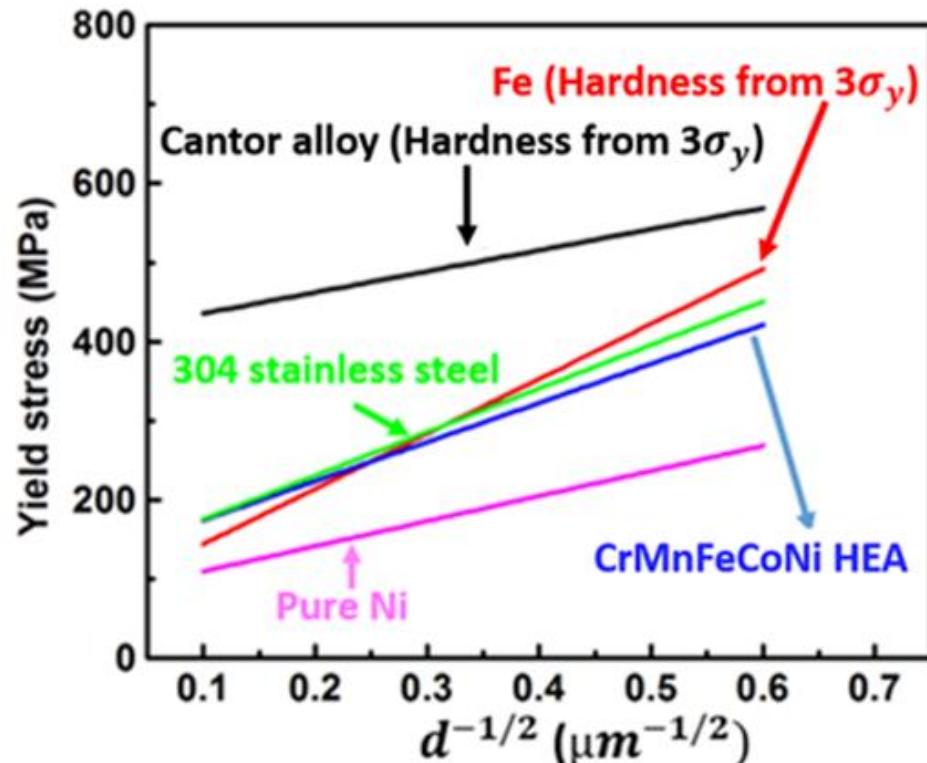
Grain size strengthening

$$\text{Growth kinetic constant: } C = A_0 \exp\left(-\frac{Q}{RT}\right)$$

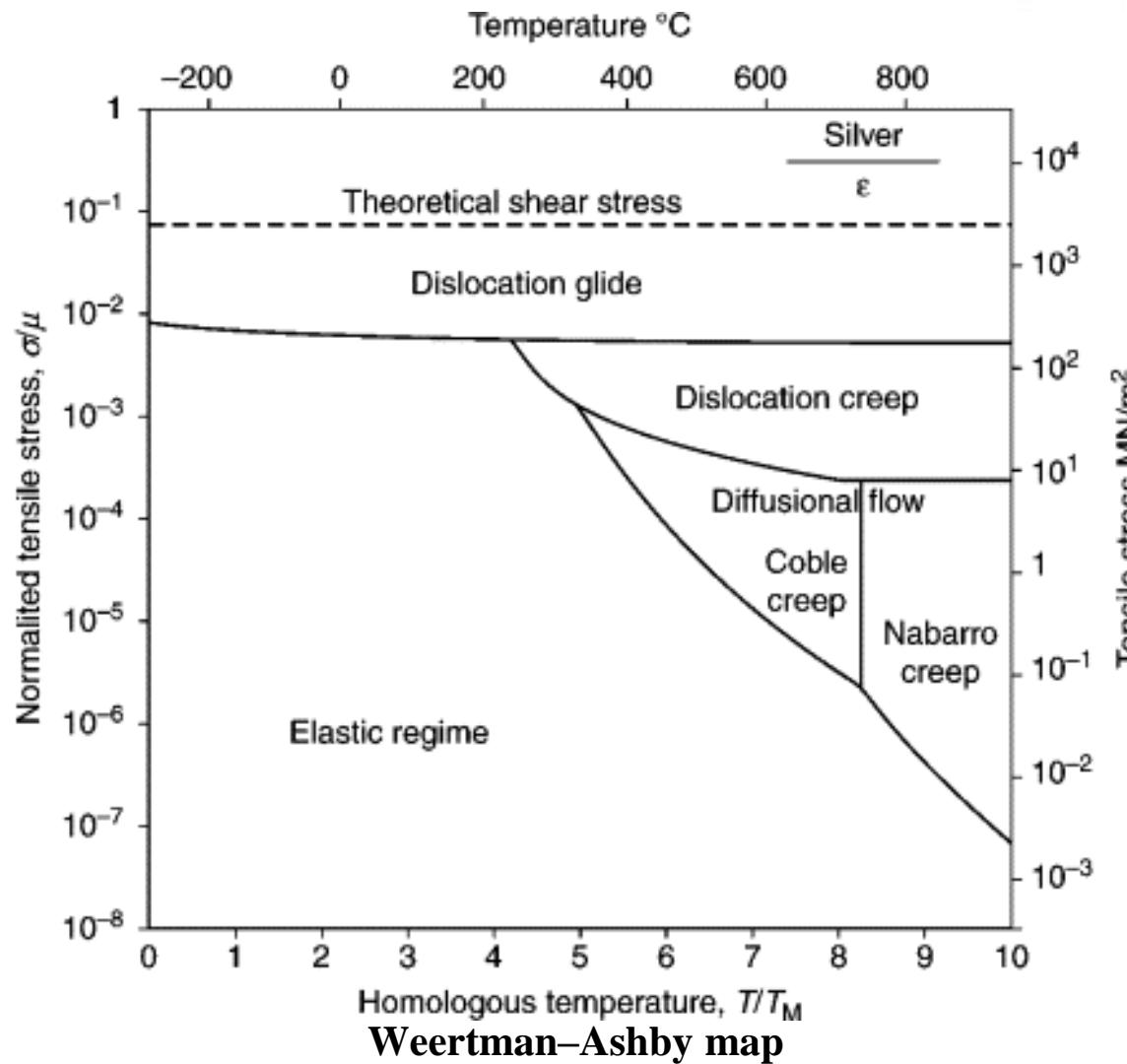
Q: activation energy for grain growth at absolute temperature T

A_0 : material constant

$$d^n - d_0^n = Ct \quad (\text{d: grain size, t: time})$$



Creep



Main creep mechanism

1. **Above the theoretical:**
shear strength: Plastic flow of the material could occur without dislocation involved.
2. $\frac{\sigma}{G} > 10^{-2}$, **Below the theoretical shear strength:** dislocation glide
3. $10^{-5} < \frac{\sigma}{G} < 10^{-2}$: Dislocation glide & climb
4. $\frac{\sigma}{G} \leq 10^{-5}$: diffusion creep

Herring-Nabarro
Vacancies migrate from one grain cell to another. (diffusion creep)

Creep

1. Transient curve vs. inverted transient curve
(stress dependence of creep mechanisms)

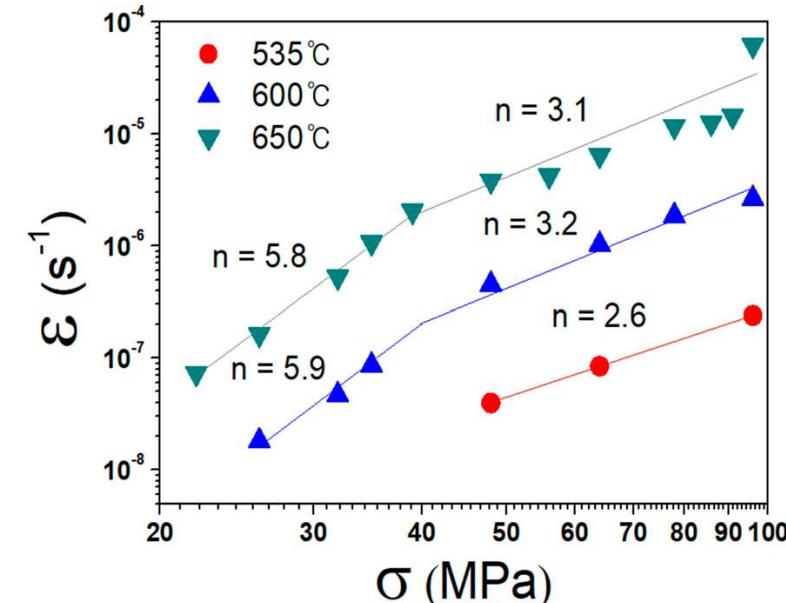
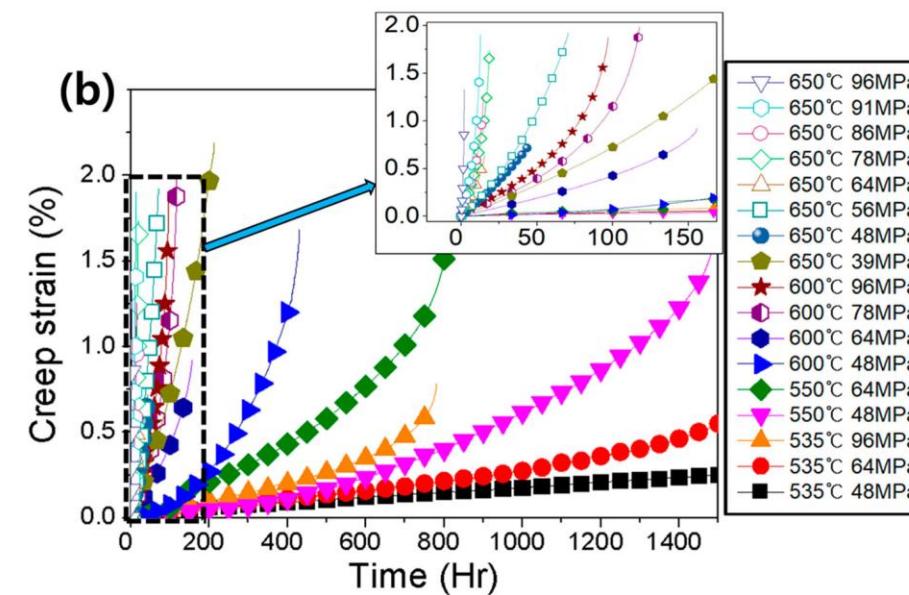
Creep stress component: $n = \partial \dot{\varepsilon}_{SS} / \partial \ln \sigma$

2. Creep rate $< 10^{-5}$

Low stress dependence (higher applied stresses)

High stress dependence (Lower applied stresses)

3. Temperature dependence



Sluggish diffusion

Lattice distortion

Stress dependence of the steady state creep rate of CoCrFeMnNi.

Creep curves of CoCrFeMnNi HEA at temperatures between 535°C and 650°C. (a) lower stresses and (b) high stresses.

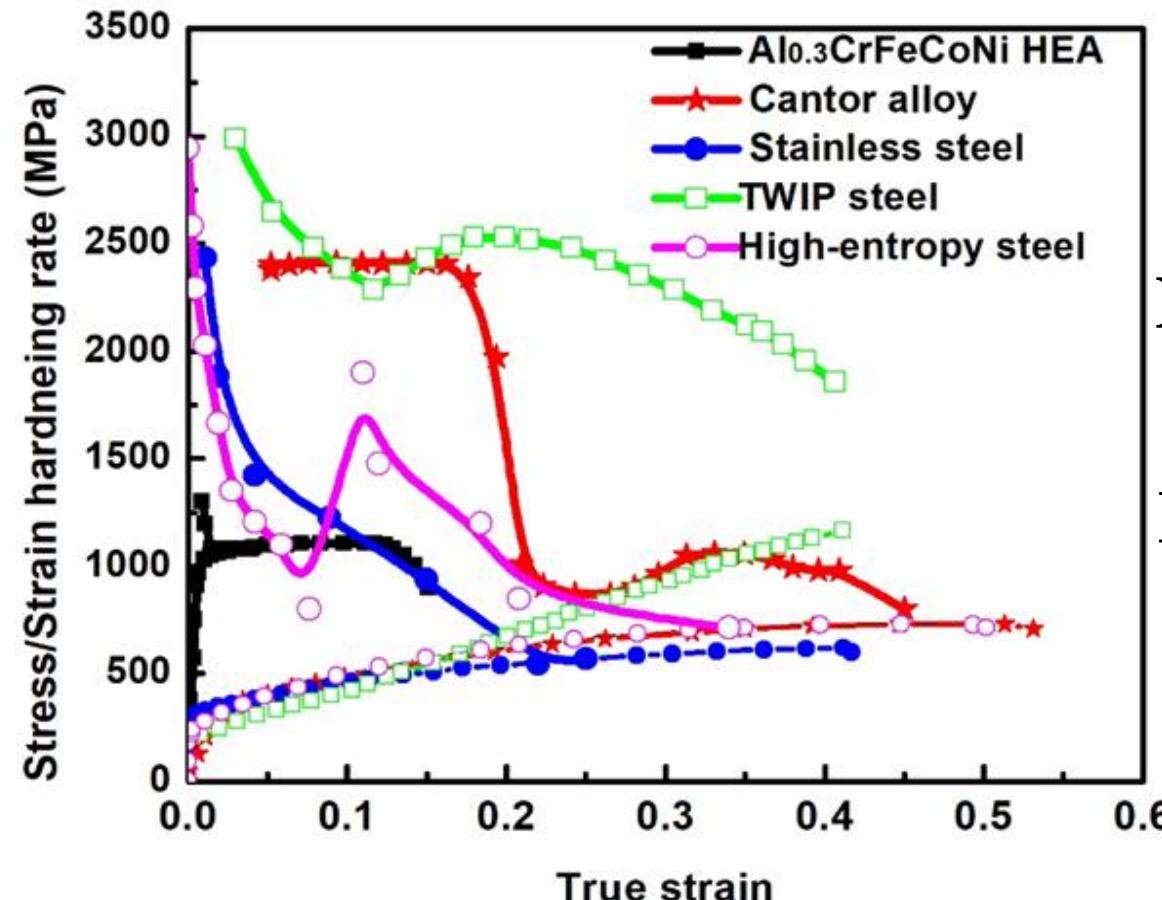
Kang, Y. B., Shim, S. H., Lee, K. H., & Hong, S. I. (2018). Dislocation creep behavior of CoCrFeMnNi high entropy alloy at intermediate temperatures. Materials Research Letters, 6(12), 689-695.

Strain hardening of HEAs

Considère criterion: $\frac{d\sigma}{d\varepsilon} = \sigma$ (Necking)

High work hardening rate, high flow stress, high toughness

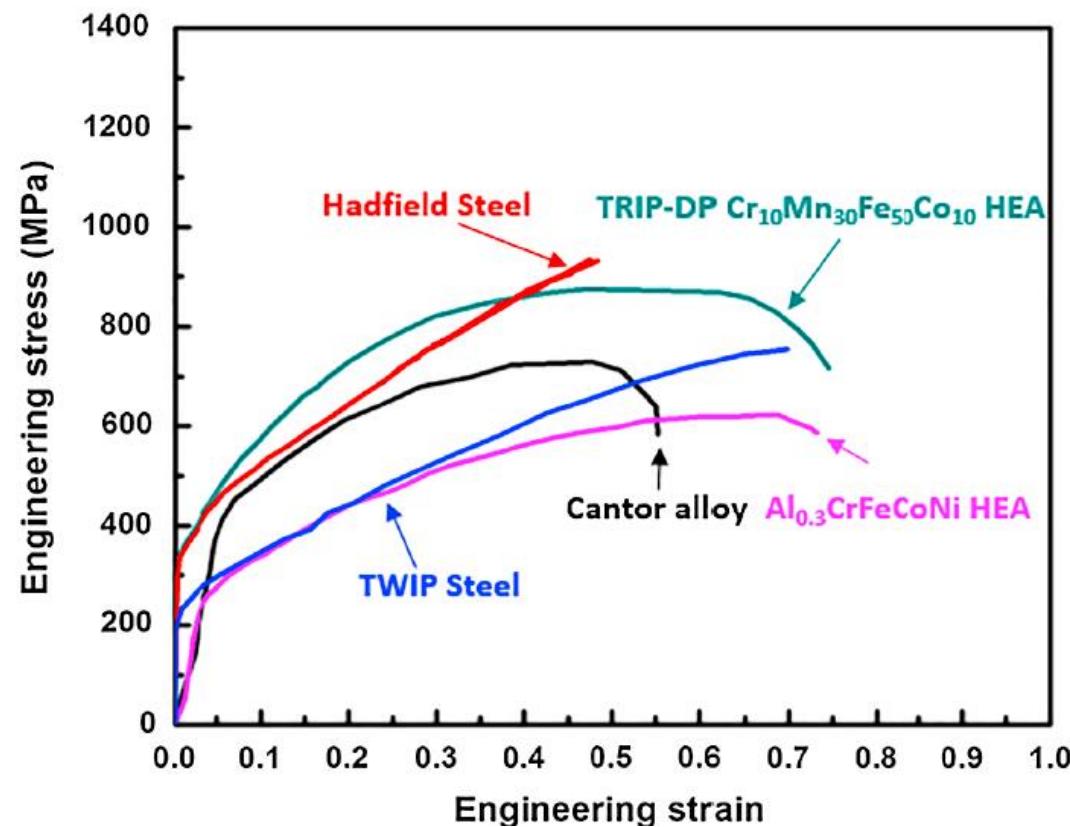
(Mechanical twinning, solid solution...)



Low $\frac{d\sigma}{d\varepsilon}$: Readily neck

High $\frac{d\sigma}{d\varepsilon}$: High toughness

Strength and ductility trade off



Twinning effect

$$\sigma_T = \alpha_T \left(\frac{\gamma_{SF}}{Gb_s} \right)^{\frac{1}{2}} \text{ (Twinning stress)}$$

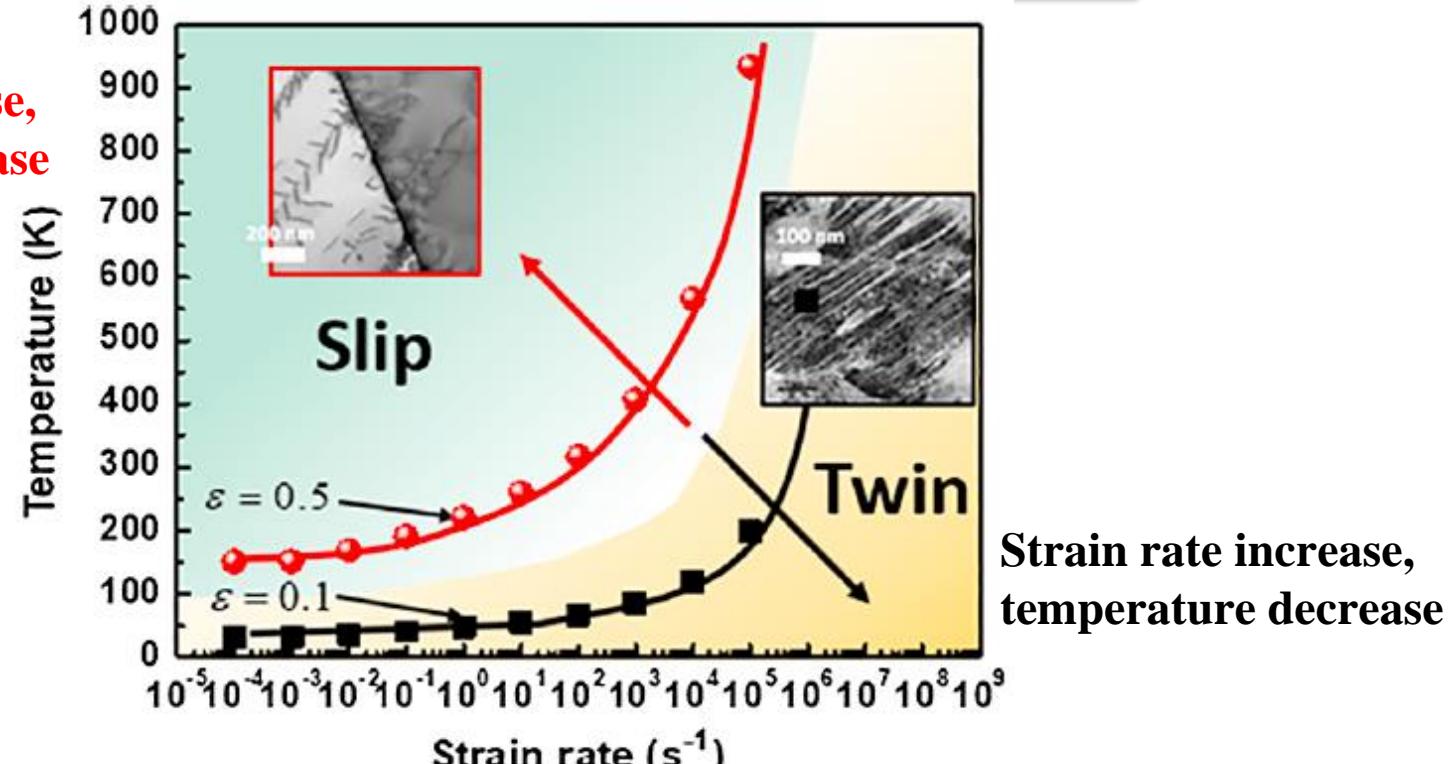
Insensitive to temperature

Zerilli-Armstrong equation: $\sigma_S = \sigma_G + C_2 \exp[-(C_3 - C_4 \ln(\frac{\dot{\varepsilon}}{\dot{\varepsilon}_0}) T] + k_s d^{-\frac{1}{2}}$

α_T : adjustable parameter
 γ_{SF} : stacking fault energy

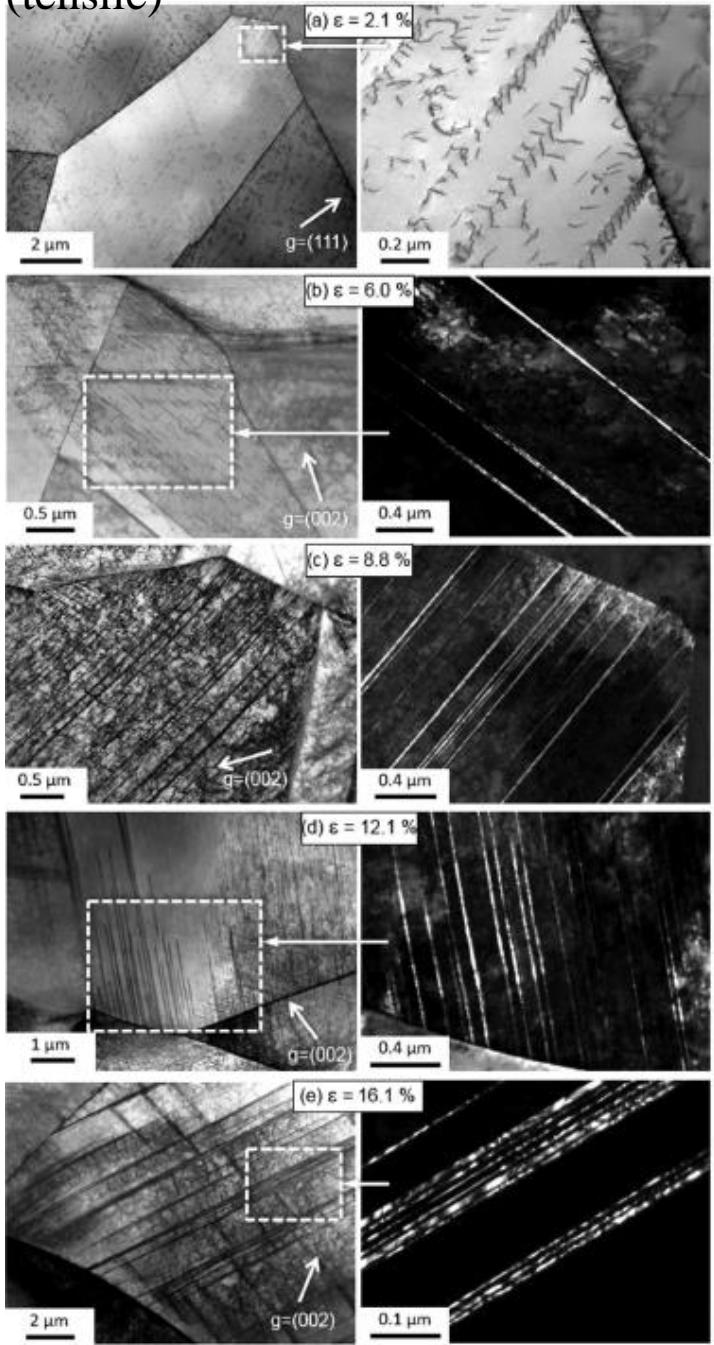
$$\gamma_{SF} \sim 25 \text{ mJ/m}^2$$
$$a_0 \sim 0.36 \text{ nm}$$

Strain rate decrease,
temperature increase



Strain rate increase,
temperature decrease

Slip and twinning regimes transition for polycrystalline CrMnFeCoNi
Cantor alloy with grain size $\sim 50 \mu\text{m}$ at two levels of plastic strain



Twinning effect

Increase in dislocation density

Initiation of nanotwins

Volume fraction of twin increases

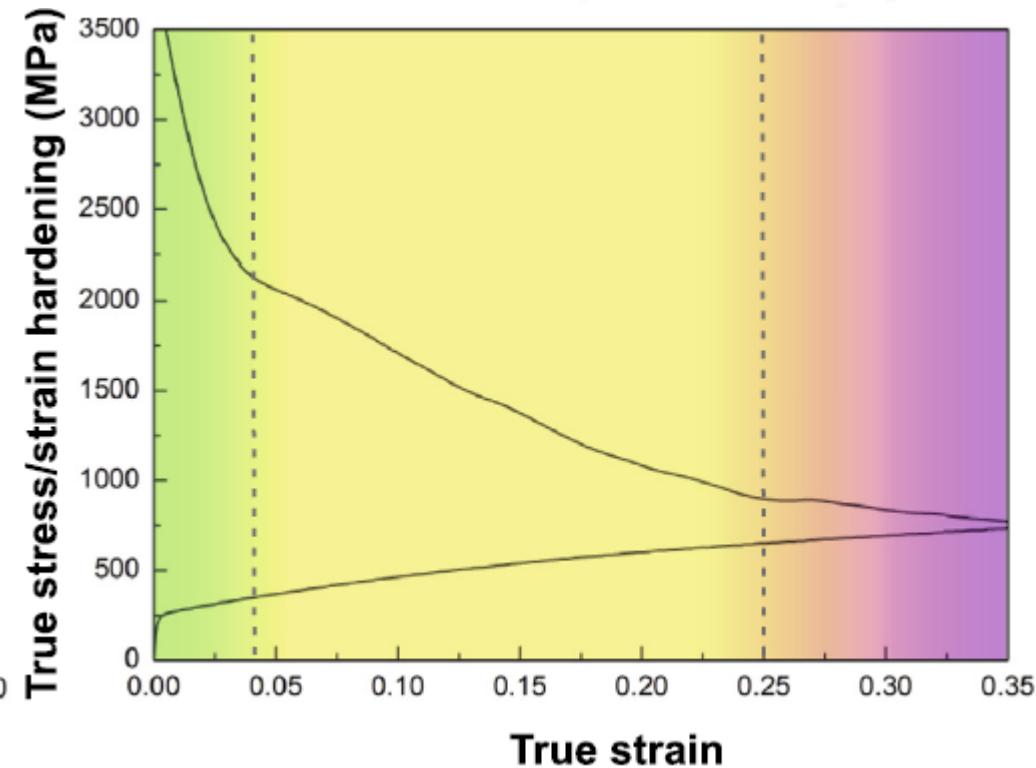
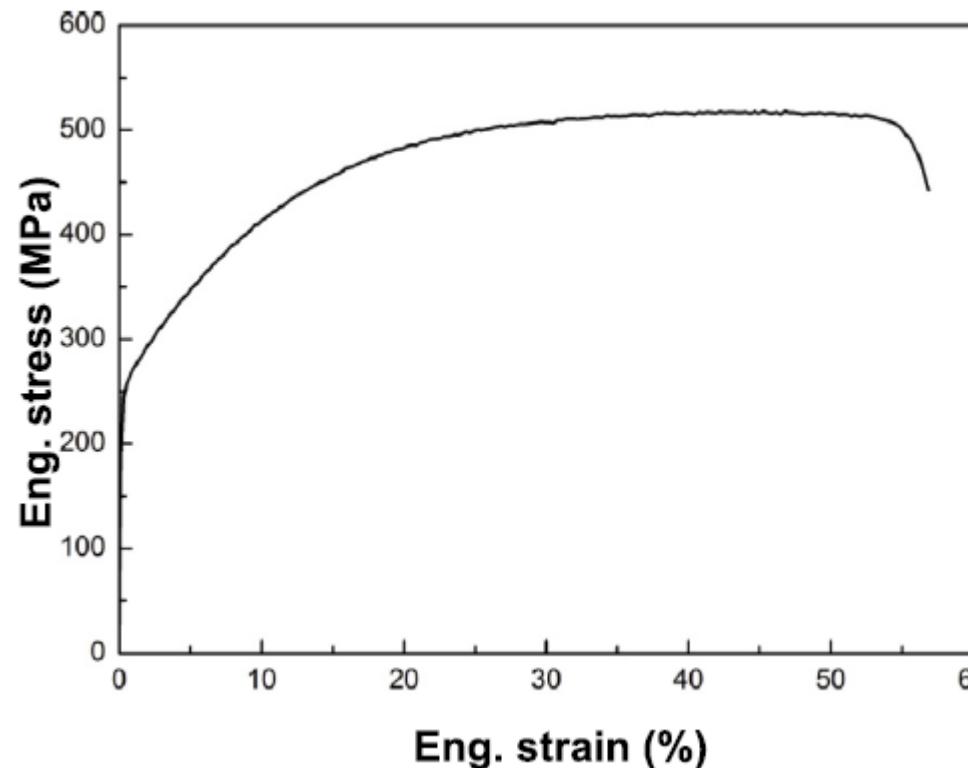
$$\sigma_T \approx 700 \text{ MPa}$$
$$\alpha_T \approx 16 \text{ GPa}$$

Intersecting twins

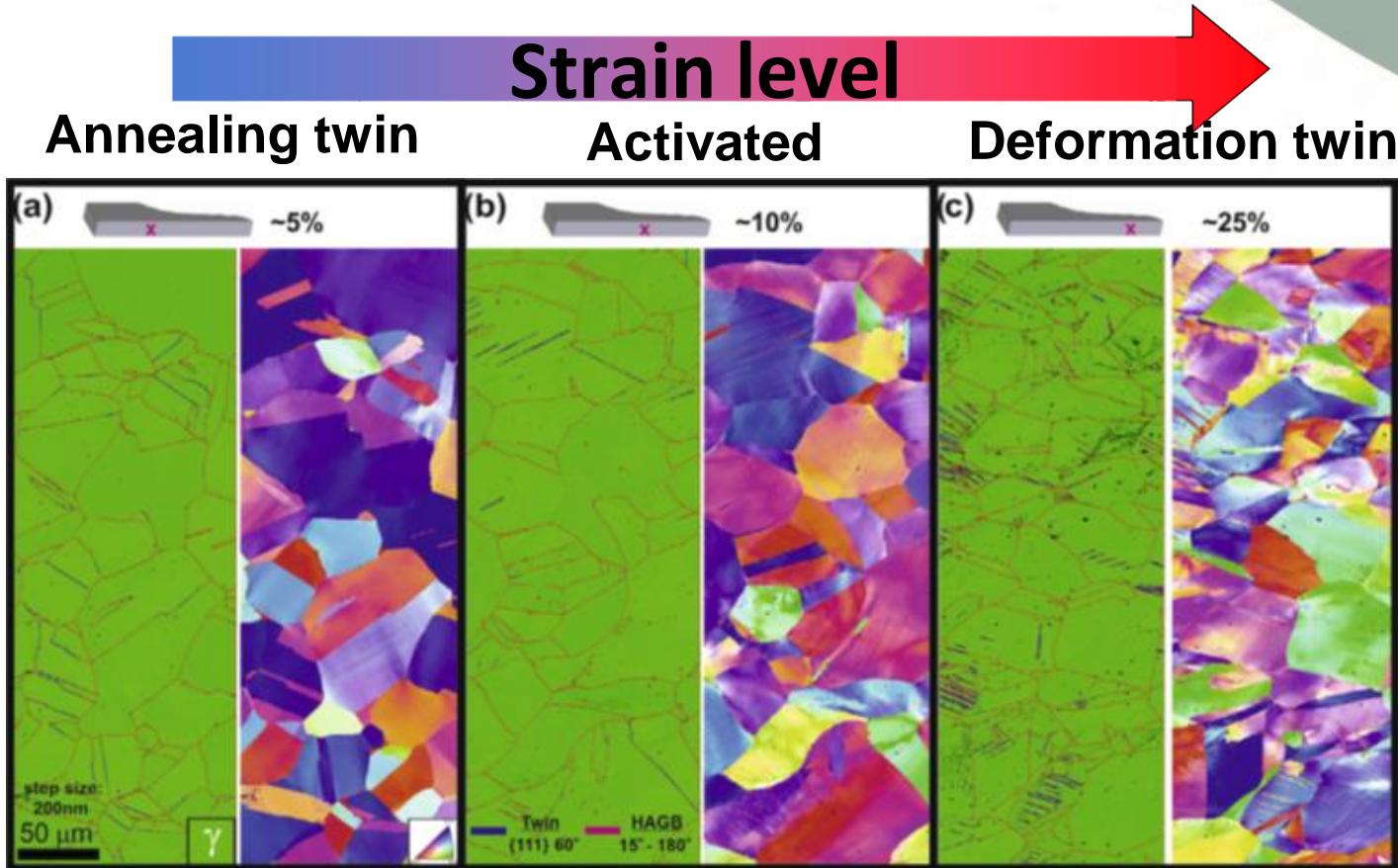
Laplanche G, Gadaud P, Horst O, Otto F, Eggeler G, George E. Temperature dependencies of the elastic moduli and thermal expansion coefficient of an equiatomic, single-phase CoCrFeMnNi high-entropy alloy. J Alloys Compd 2015;623:348–53.

Twinning-induced plasticity (TWIP)

1. Major hardening mechanism of stainless steel
2. Twinning does not contribute primary effect at room temperature tensile deformation in Cantor alloy.
3. Reducing Ni to lower SFE with four component system.
4. Reducing Co, Cr contents.



Twinning-induced plasticity (TWIP)

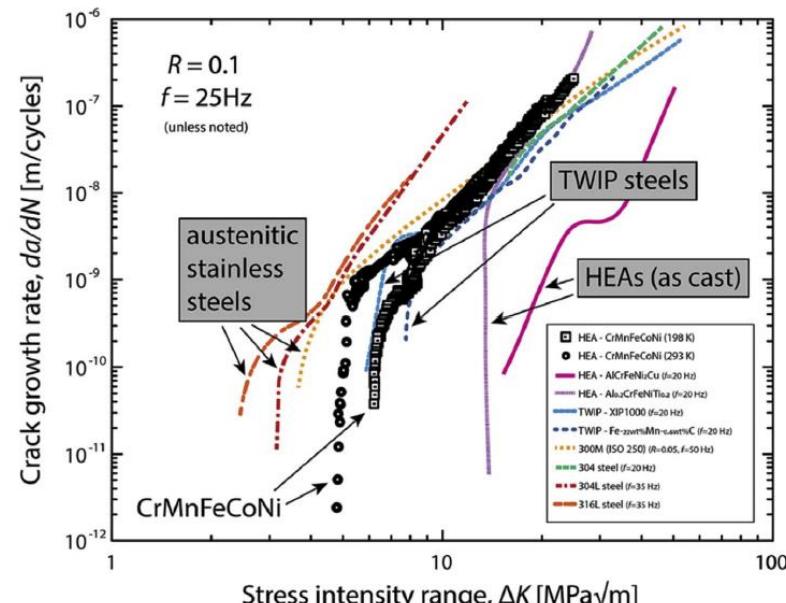
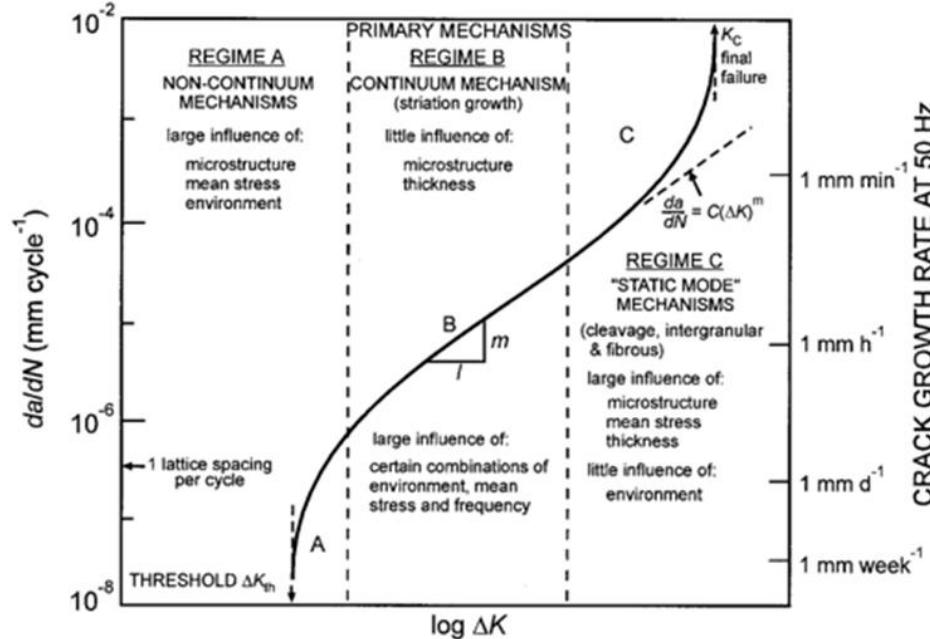


1. Deformation twin activated (~10%)
2. Work hardening by impeding movement of dislocation
3. Strain level increases, fraction of twin boundaries increase

EBSG and IPF images at cross section of the deformed $\text{Fe}_{40}\text{Mn}_{40}\text{Co}_{10}\text{Cr}_{10}$ sample at varied strain level

Twin

HAGB



Fatigue

Fatigue crack growth:

$$\frac{da}{dN} = f[\Delta K, K_{max}(\text{or } R), v, \text{environment}, \text{waveform} \dots]$$

Stress intensity range:

$$\Delta K = K_{max} - K_{min}$$

$$\frac{da}{dN} = C(\Delta K)^m$$

C & m: material scaling constant (2~4 for metal)

K: stress intensity

Four stages of fatigue:

- Initial cyclic damage (cyclic softening or hardening)
- Generation of initial flaw
- Macroscopic propagation of flaw
- Catastrophic failure

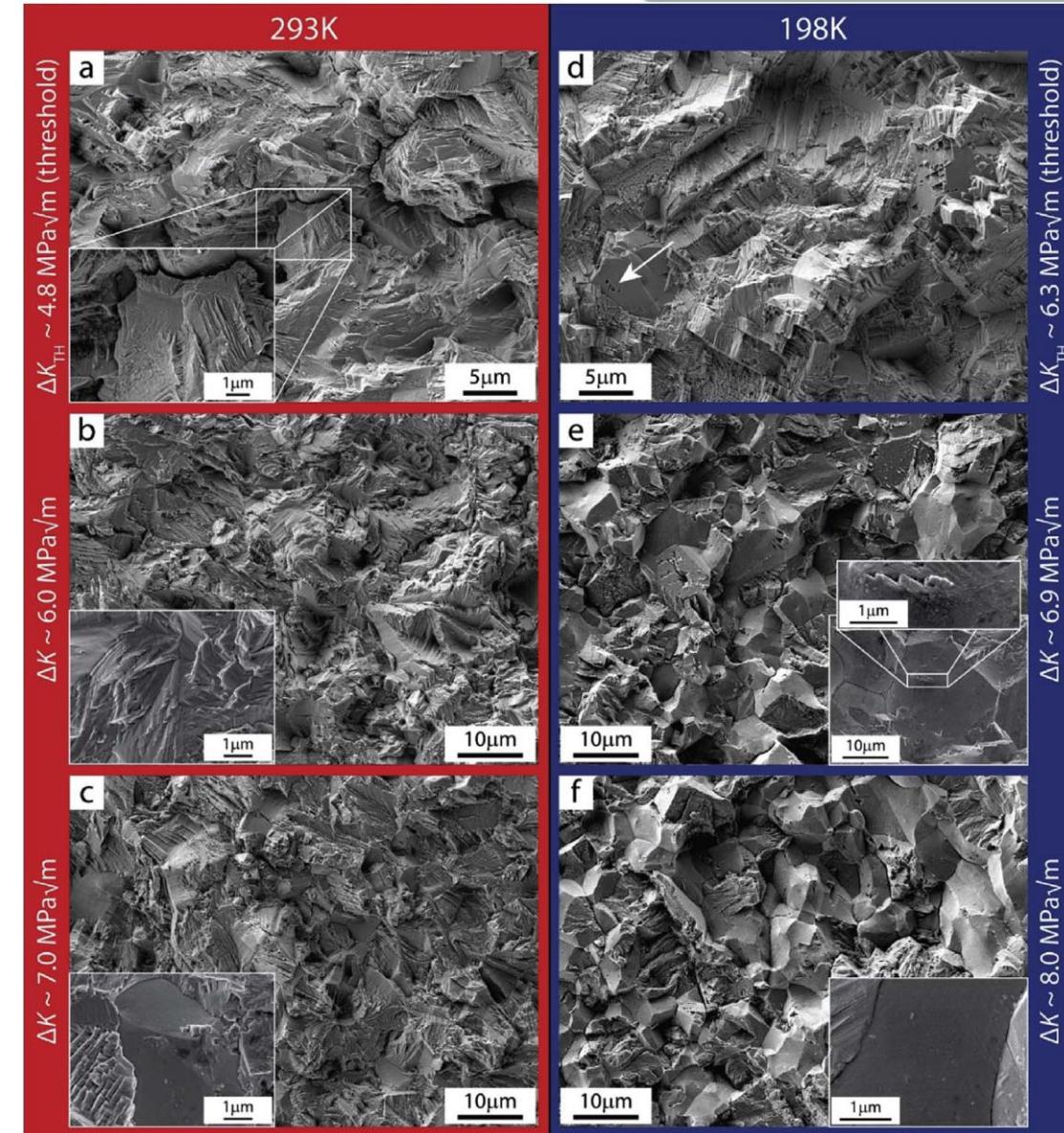
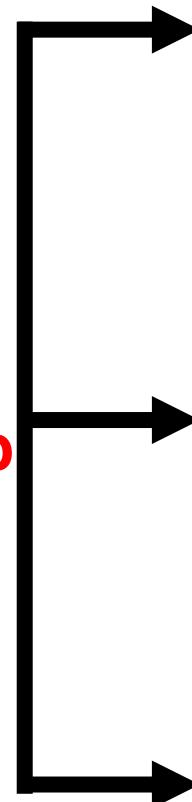
1. Ritchie RO. Influence of microstructure on near-threshold fatigue-crack propagation in ultra-high strength steel. Met Sci 1977;11:368–81.

2. Thurston KV, Gludovatz B, Hohenwarter A, Laplanche G, George EP, Ritchie RO. Effect of temperature on the fatigue-crack growth behavior of the high-entropy alloy CrMnFeCoNi. Intermetallics 2017;88:65–72.

Fatigue 2

Fractographic analysis of CrMnFeCoNi samples
tested at 293K and 198K.

No twins
Planar slip

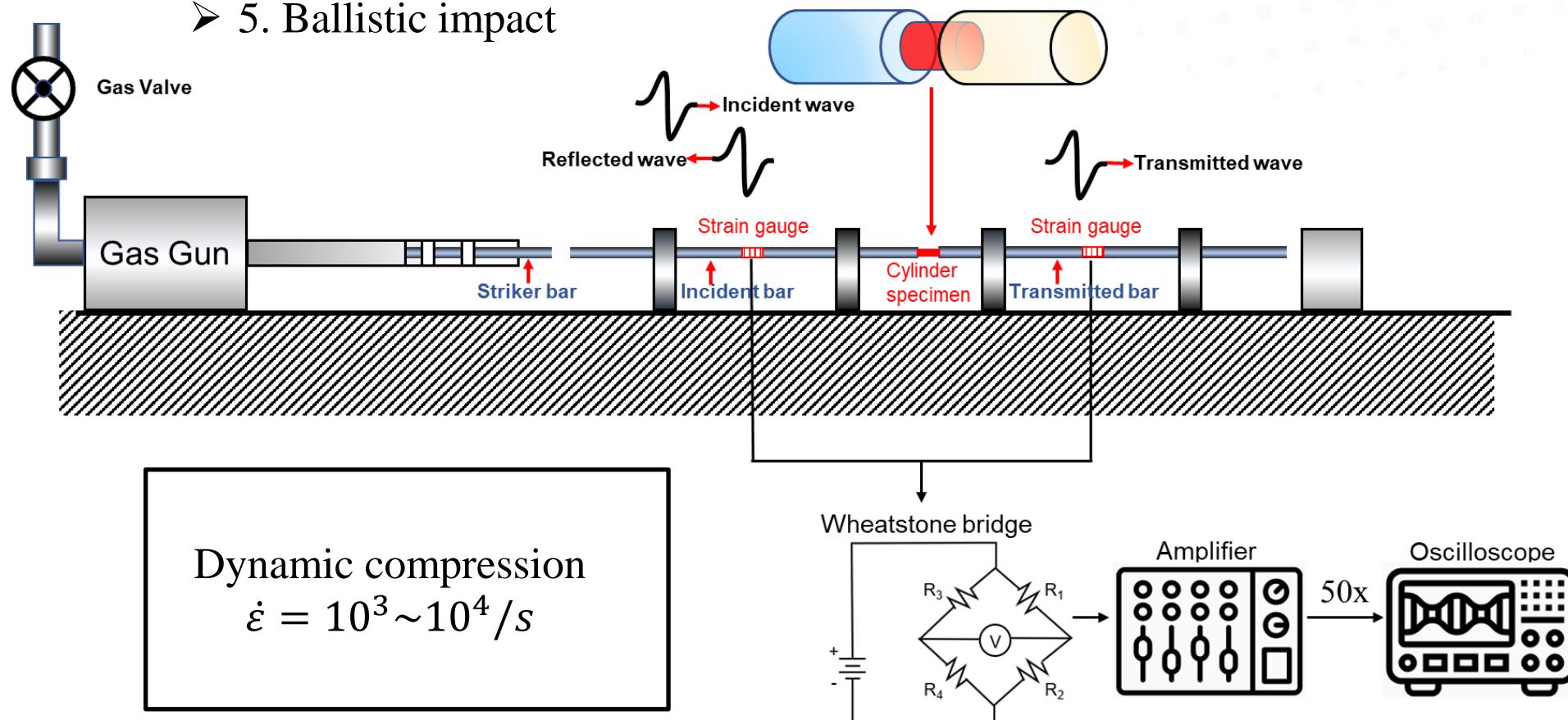


Planar slip

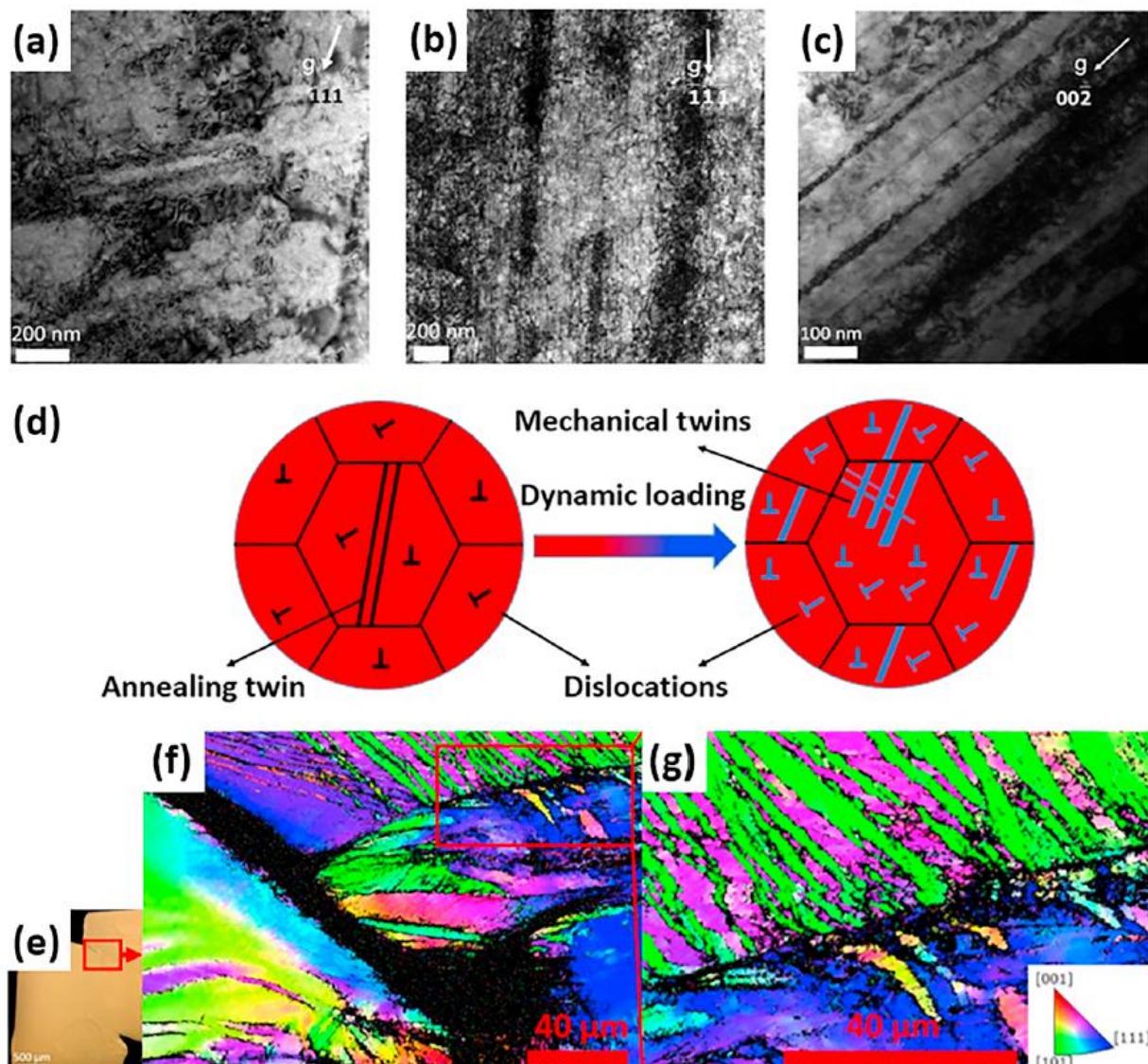
1. ΔK_{TH} (fatigue threshold)
Cracks won't initiate below the fatigue threshold
2. Twins

High-strain rate and high-pressure deformation

- 1. Hopkinson bar experiment
- 2. Gas gun impact experiment
- 3. Wave reflection experiment (ultimate tensile strength)
- 4. Shear localization experiment (failure by adiabatic shear bands)
- 5. Ballistic impact



Dynamic properties of HEA



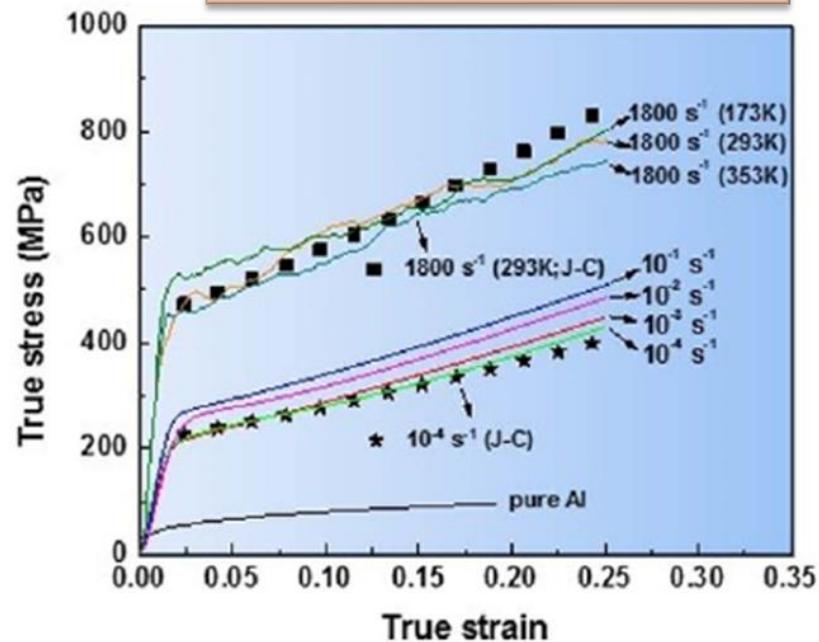
TEM bright-field images of the deformed samples at the strain rates of (a) 10^{-4} s^{-1} ; (b) and (c) 1800 s^{-1} .

Solid solution

Multiple strengthening effect

Forest dislocation

Mechanical twinning

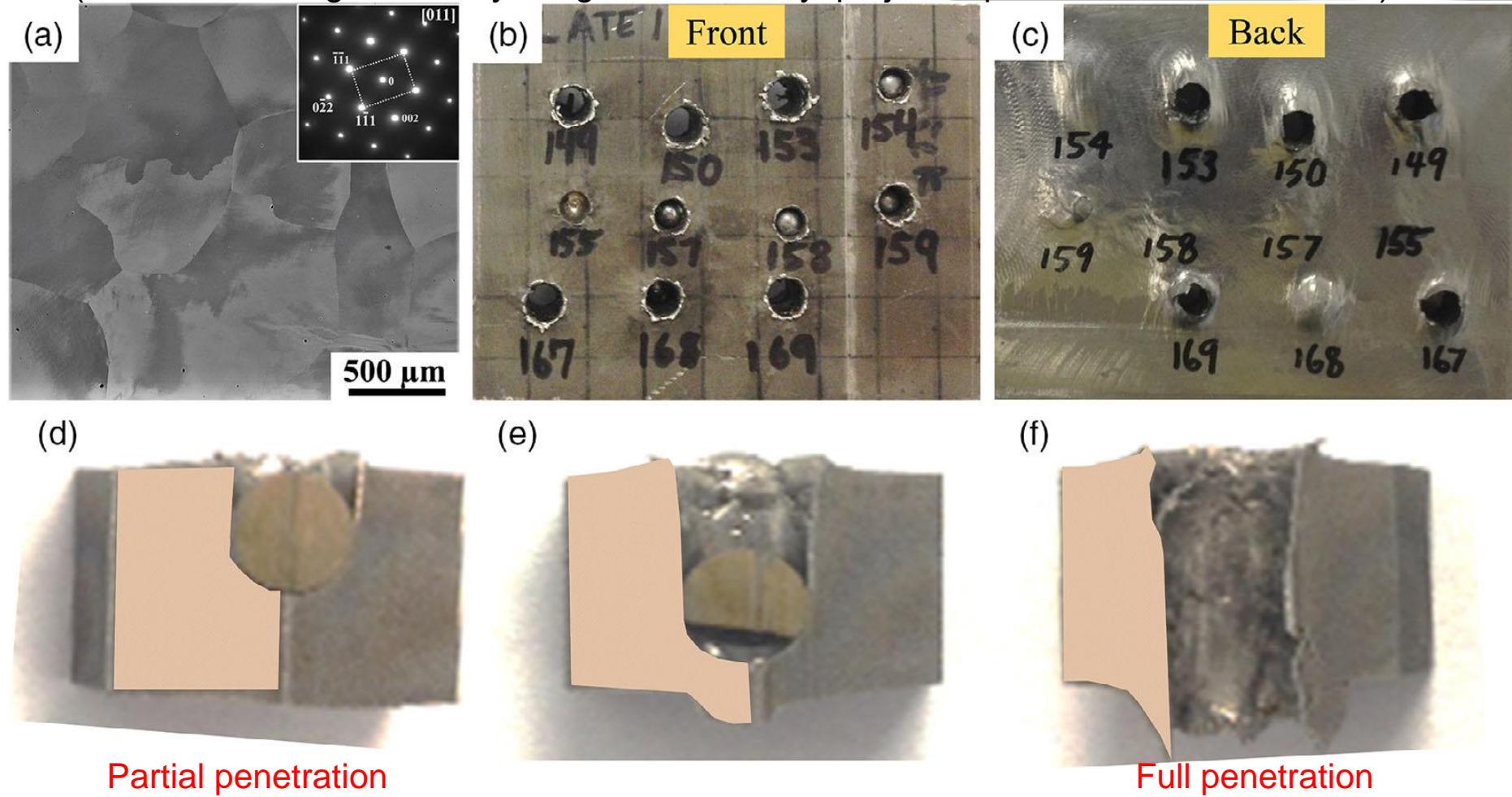


Ballistic performance

Understanding dynamic behavior of HEA at range of $10^3\text{-}10^5\text{s}^{-1}$

(thickness, strength, ductility, toughness, density, projectile parameter, local strains...)

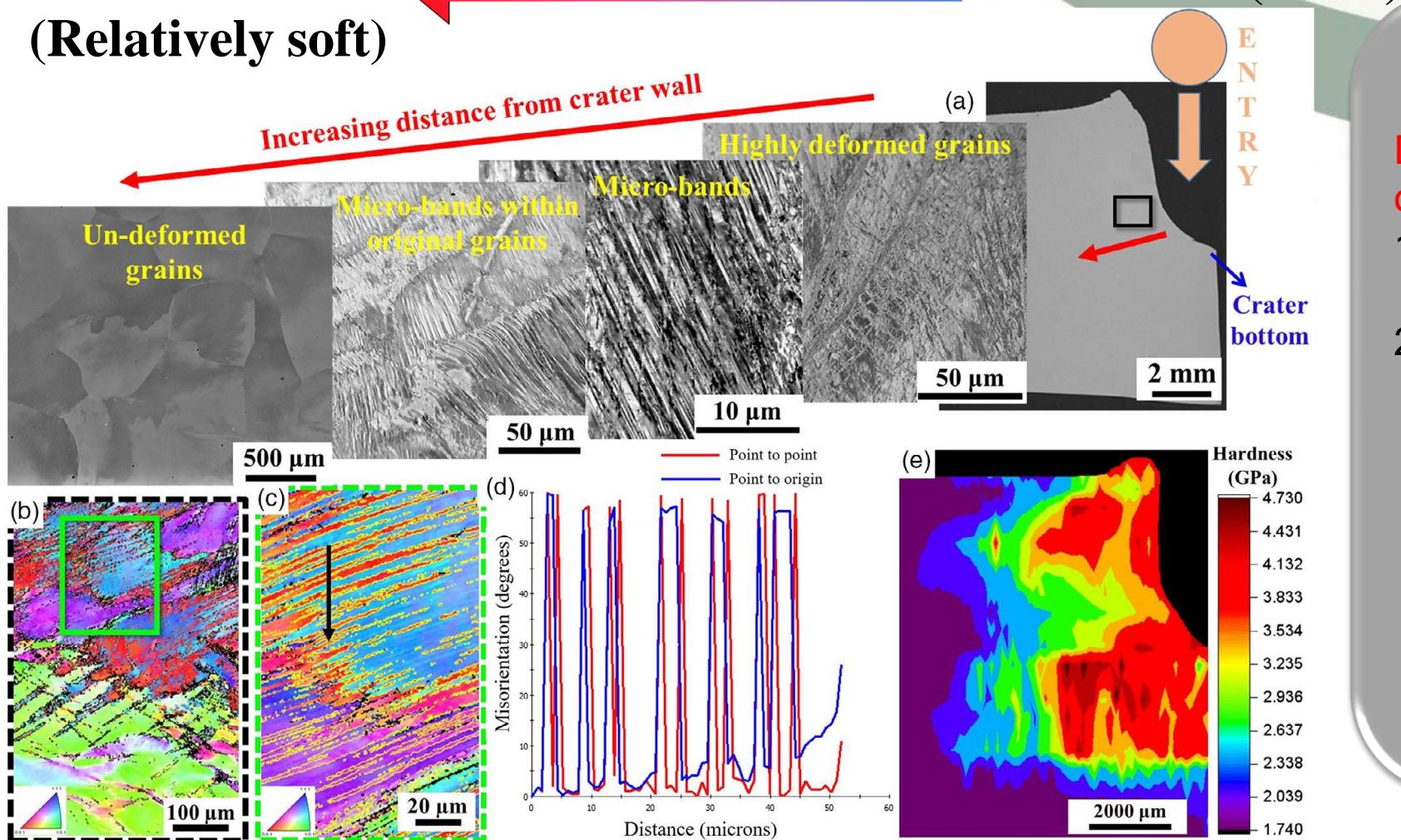
E52100 steel
(RC60)
projectiles



(a) Microstructure of as-cast Al0.1CoCrFeNi HEA (b) front and back side of as-cast HEA plate after ballistic test Penetration at velocity (c) 594m/s (d) 739m/s (f) 994m/s

Ballistic performance

Inner structure
(Relatively soft)



(a) Microstructure as a function of distance from the crater wall (b)(c)IPF map
(d) Misorientation profile (e) nano-indentation hardness contour map

Micoband and micro-twinning formation

Due to spherical shock wave on Low SFE metals:

1. Misorientation profile: ~60°(twinning)
2. Misorientation profile: <10°(Micoband)

Work hardening

1. No dynamic recrystallization caused by ASB
2. Excellent work-hardening at crater wall

Conclusion

- 1. High thermal stability
- 2. High fracture toughness
- 3. Excellent wear resistance
- 4. Outstanding working hardening ability
- 5. Appreciable anti-fatigue and anti-creep ability
- 6. Excellent ballistic impact performance

Future direction

- 1. Light novel high-entropy alloy
- 2. Non-metallic element doped high-entropy alloy
- 3. Wrought processed high-entropy alloy (homogenization, cold rolling/hot rolling)
- 4. BCC-HEA

Challenges

- 1. Formation of different phase
- 2. Identifying useful alloy composition
- 3. Expensive